

# Variational phase-field models of fracture.

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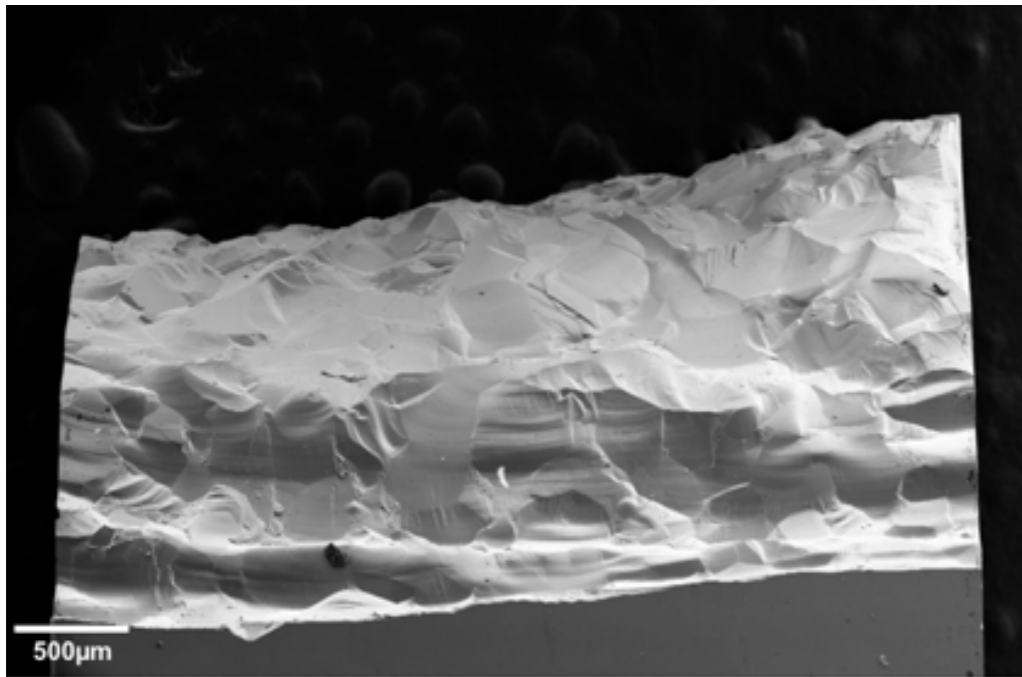
Source code: <http://bitbucket.org/bourdin/mef90-sieve>

dockerhub: bourdin/mef90-mpich

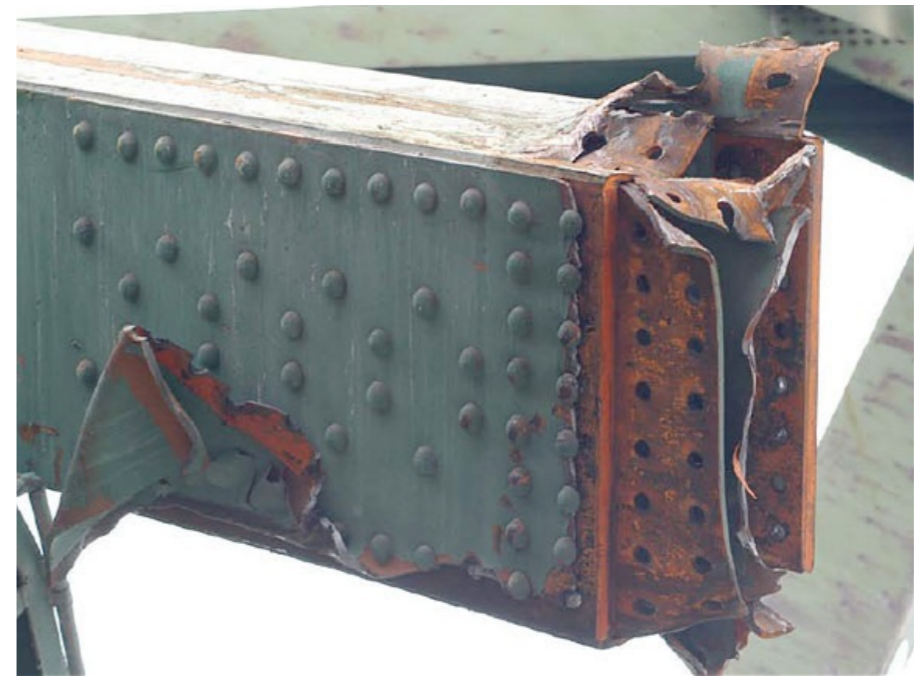
NSF DMS-0908267, DMS-0605320, DMS-1312739, DMS-1716763

LA Board of Regents, Chevron ETC, Corning Inc., Asahi Glass Company

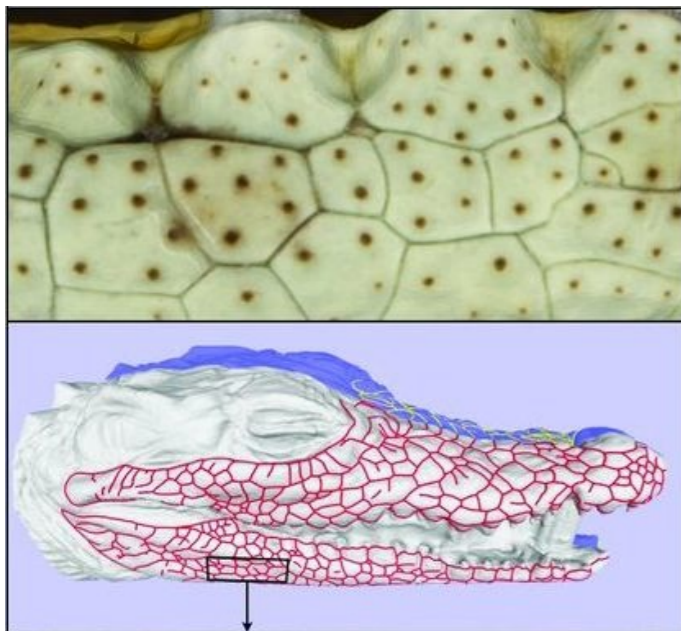
# Fracture Mechanics



Fracture surface of AlON



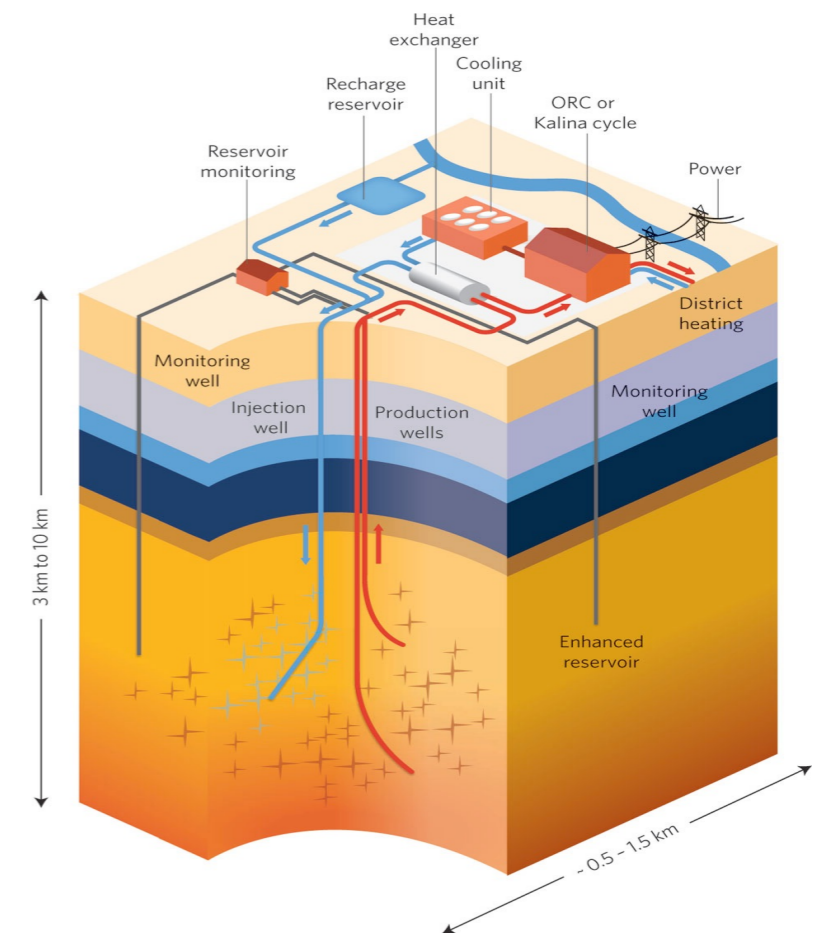
Beam of the I-35W bridge, Minneapolis



Crocodile skin



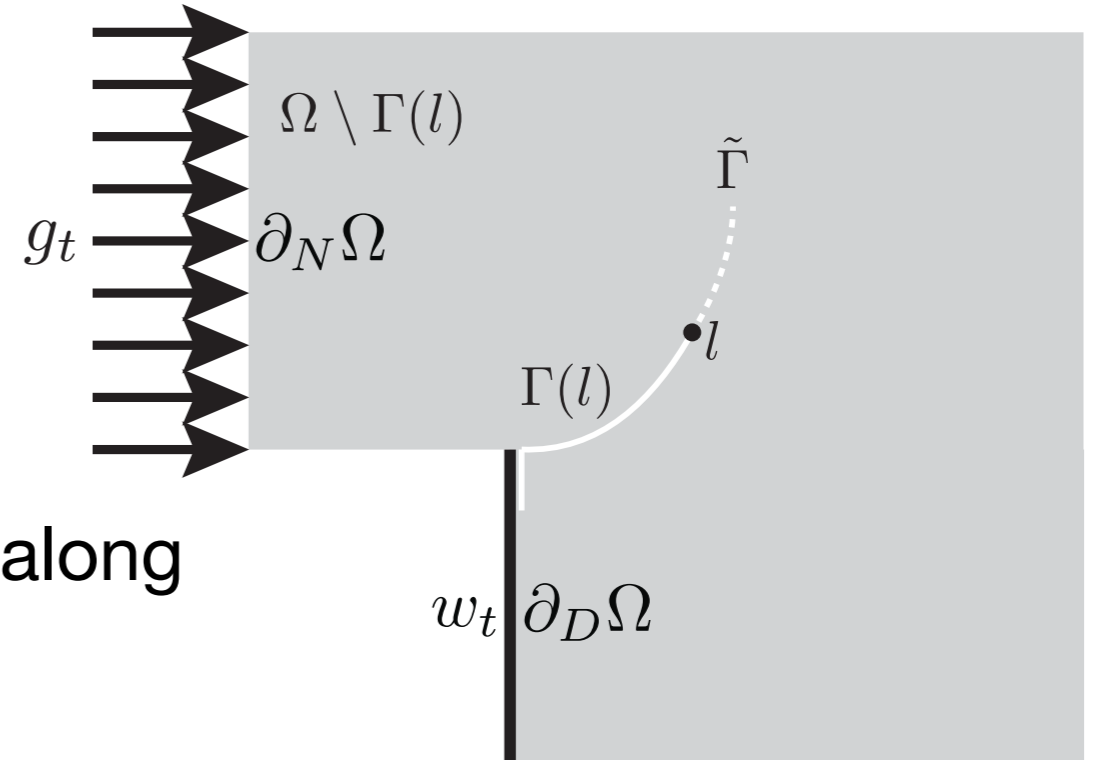
Causeway of Giants



Enhanced Geothermal Systems

# Revisiting Griffith's argument

- Quasi-static, loading parameter (“time”)  $t$
- Boundary displacement  $u(t) = w_t$  on  $\partial_D\Omega$
- Boundary and body forces  $f_t, g_t$ .
- Single crack  $\Gamma(l)$  with length  $l$  propagating along a known path in 2D
- Equilibrium displacement  $u(t, l)$
- Potential energy = mechanical energy of equilibrium displacement.



$$\mathcal{P}(t, l) := \mathcal{E}(t; l, u(t, l))$$

# Revisiting Griffith's argument

## Griffith argument:

- Crack growth hypothesis:  
 $l(t)$  non-decreasing function of  $t$
- Fracture energy proportional to crack length:  
 $G_c$ : rate of energy per unit of length (area)
- Stability principle: for any  $\Delta l \geq 0$

$$\mathcal{P}(t, l(t) + \Delta l) + G_c \Delta l \geq \mathcal{P}(t, l(t))$$

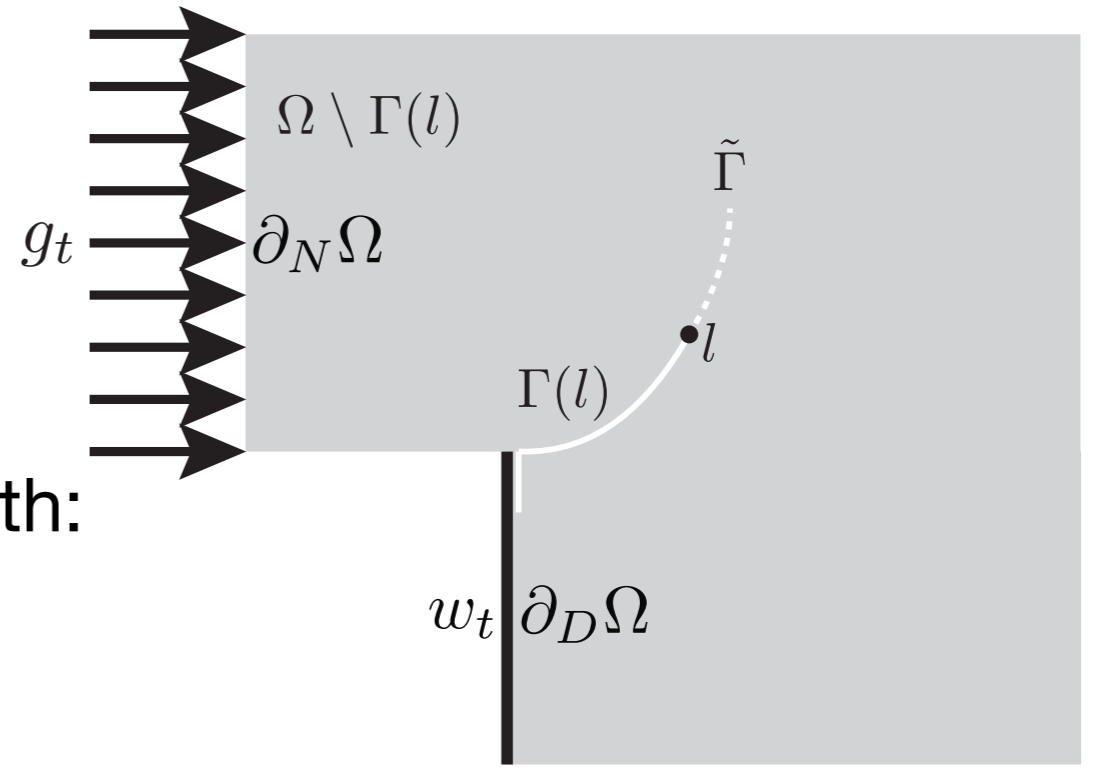
$$G(t, l(t)) := -\frac{\partial \mathcal{P}}{\partial l}(t, l(t)) \leq G_c$$

- Stability principle: if  $\dot{l}(t) \geq 0$ , for any  $\Delta l \leq 0$

$$\mathcal{P}(t, l(t) - \Delta l) - G_c \Delta l \geq \mathcal{P}(t, l(t))$$

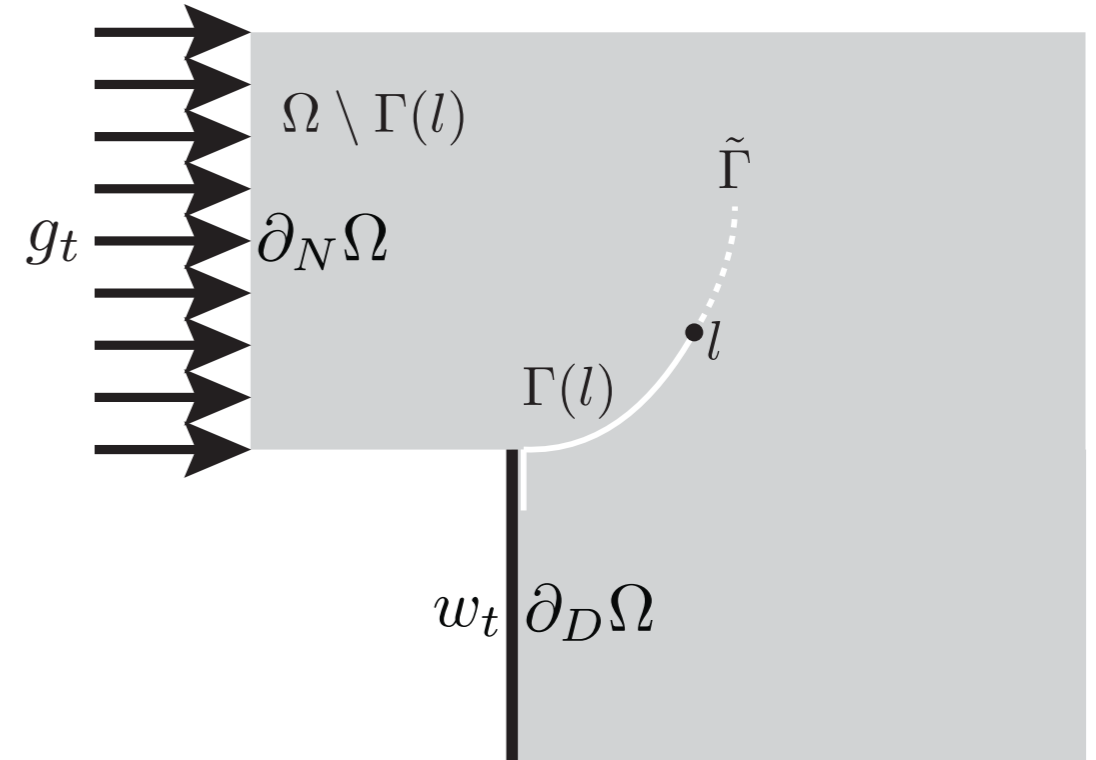
$$-\frac{\partial \mathcal{P}}{\partial l}(t, l(t)) \geq G_c$$

$$\left\{ \begin{array}{l} \dot{l}(t) \geq 0 \\ G(t, l(t)) \leq G_c \\ (G(t, l(t)) - G_c) \dot{l}(t) = 0 \end{array} \right.$$



# A variational view of Griffith theory

$$\begin{cases} \dot{l}(t) & \geq 0 \\ G(t, l(t)) & \leq G_c \\ (G(t, l(t)) - G_c) \dot{l}(t) & = 0 \end{cases}$$



- Griffith solutions are minimizers of  $\mathcal{P}(t, l(t)) + G_c l(t)$  under a crack growth constraint.
- Variational view of Griffith at all  $t$ , find  $(u(t), l(t))$  minimizing

$$\mathcal{E}(t, v, l) := \int_{\Omega \setminus \Gamma(l)} \frac{1}{2} \mathbf{A} e(v) \cdot e(v) dx - \mathcal{L}(t; v) + G_c l$$

amongst all  $l$  such that  $l > l(s)$  for all  $s < t$ ,  $v$  admissible.

# Francfort-Marigo's variational model

- Griffith energy in variational form

$$\mathcal{E}(t, u, l) := \int_{\Omega \setminus \Gamma(l)} \frac{1}{2} \mathbf{A} e(u) \cdot e(u) dx - \mathcal{L}(t; u) + G_c l$$

- Generalized energy for *any* path:

$$\mathcal{E}(t, u, \Gamma) := \int_{\Omega \setminus \Gamma} \frac{1}{2} \mathbf{A} e(u) : e(u) dx - \mathcal{L}(u, t) + G_c \mathcal{H}^{n-1}(\Gamma)$$

- Francfort-Marigo '98: at each  $t$ , find  $u, \Gamma$  minimizing  $\mathcal{E}$  under crack growth and energy balance constraints (unilateral minimality)

“The “theorem of minimum energy” may be extended [...] if account is taken of the increase of surface energy which occurs during the formation of cracks.” - Griffith 1921

# Francfort and Marigo's variational approach

- Total fracture energy consistent with that of Griffith in variational form

$$\mathcal{E}(t; u, \Gamma) = \int_{\Omega \setminus \Gamma} \frac{1}{2} \mathbf{A} e(u) \cdot e(u) dx - \mathcal{L}(t; u) + G_c \mathcal{H}^{n-1}(\Gamma)$$

- Crack path is part of the minimization problem (Free Discontinuity problem).
- No hypotheses on crack geometry, (including connectivity, topology etc).
- Solution consistent with Griffith in variational form.
- No forces!
- Can be extended to account for:
  - Unilateral contact;
  - Non homogeneous, non-isotropic materials;
  - Non-linear elasticity;
  - ...

# Variational Phase-Field models

Generalized Ambrosio-Tortorelli functional:

$$\mathcal{E}_\ell(u, \alpha) := \int_{\Omega} a(\alpha) W(e(u)) dx + \frac{G_c}{4c_w} \int_{\Omega} \frac{w(\alpha)}{1^\ell} + \ell |\nabla \alpha|^2 dx$$
$$a(0) = 1, a(1) = 0, w(0) = 0, w(1) = 1, c_w = \int_0^1 \sqrt{w(s)} dx$$

$\Gamma$ —convergence to Francfort-Marigo variational energy:

- Static scalar problem: B '98, Braides '98:
- Finite element approximation if  $h \ll \ell$ : Bellettini-Coscia '94, B '98, Ortner-Burke-Sülli '10, '15.
- Quasi-static evolution: Giacomini '05:
- Linearized elasticity: Chambolle '04, '06, Iurlano '13, '14, '18.

MANY extensions, implementations

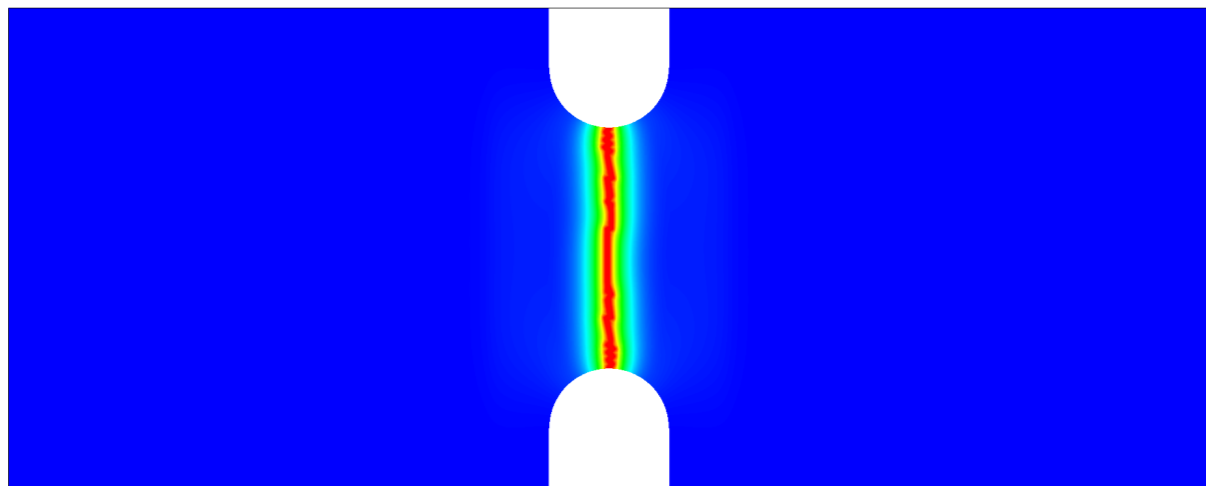


# AT1 and AT2 models

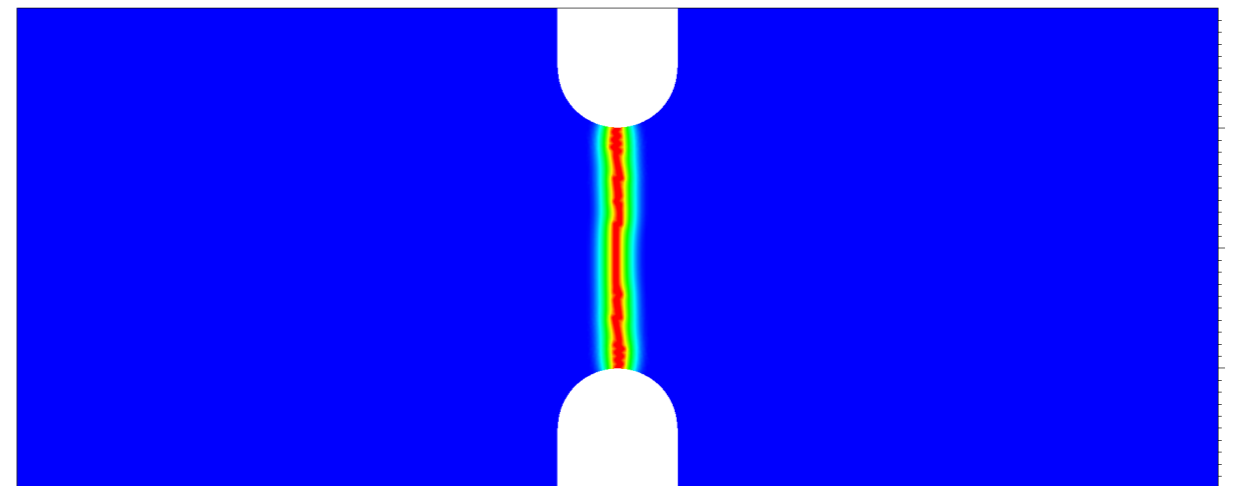
$$\text{AT2: } \mathcal{E}_\ell(u, \alpha) = \int_{\Omega} (1 - \alpha)^2 W(e(u)) dx + \frac{G_c}{2} \int_{\Omega} \frac{\alpha^2}{\ell} + \ell |\nabla \alpha|^2 dx$$

$$\text{AT1: } \mathcal{E}_\ell(u, \alpha) = \int_{\Omega} (1 - \alpha)^2 W(e(u)) dx + \frac{3G_c}{8} \int_{\Omega} \frac{\alpha}{\ell} + \ell |\nabla \alpha|^2 dx$$

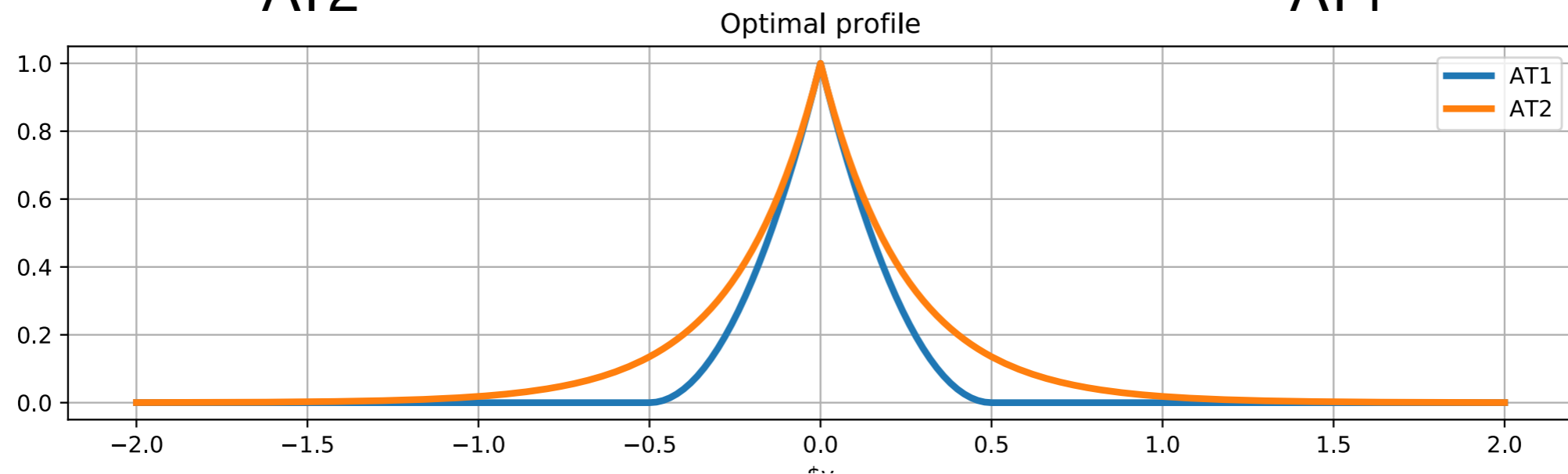
Construction of *optimal profile* as part of  $\Gamma$ -convergence *recovery sequence*.



AT2



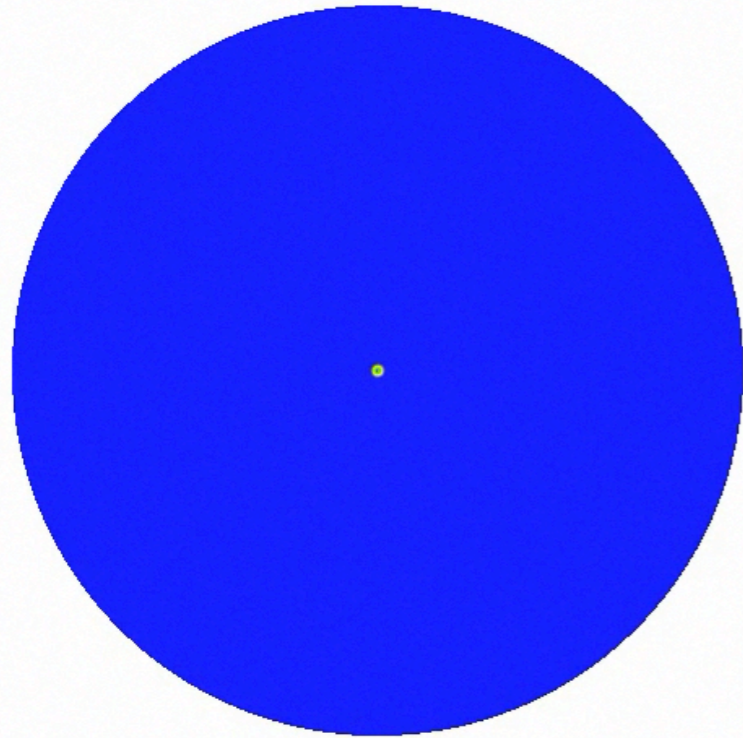
AT1



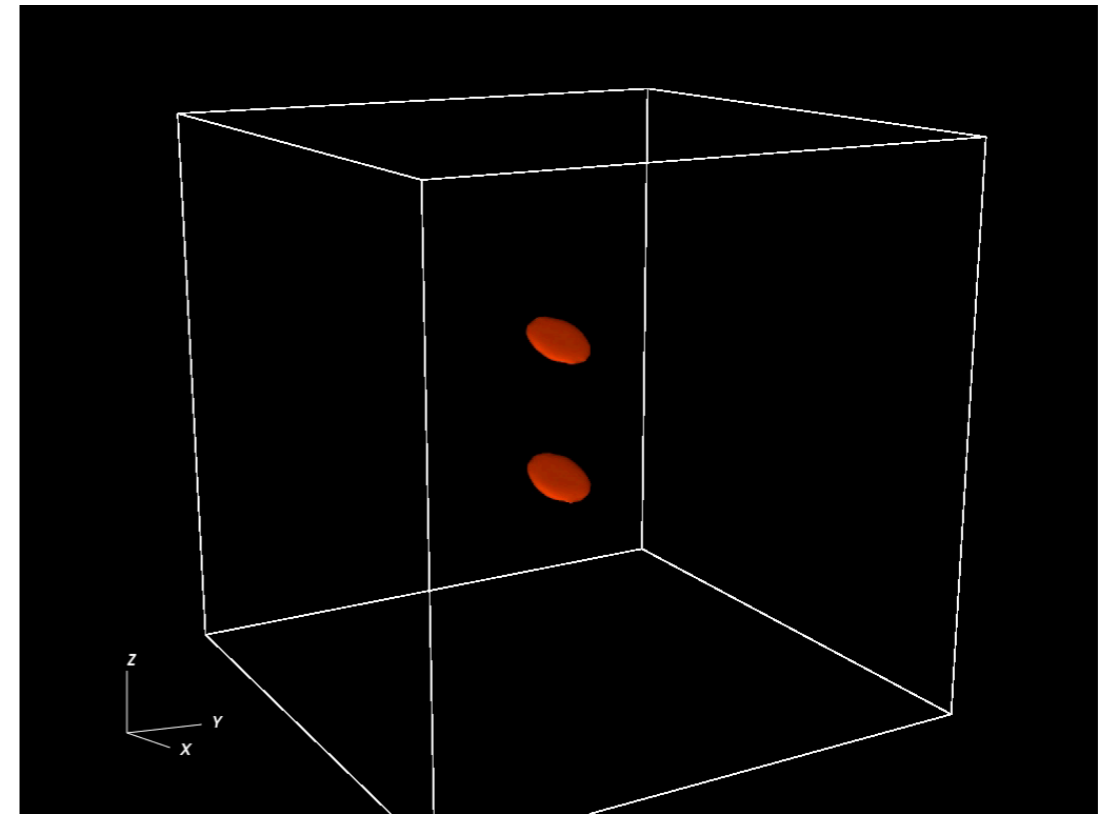
# Numerical Implementation

- Time discrete alternate minimization algorithm.
  - At each time step: *iterate* until convergence
    - Minimization w.r.t  $u$  (elastic equilibrium).
    - Constrained minimization w.r.t  $\alpha$  (variational inequality).
  - Globally stable, convergence to a critical point of the regularized energy, *monotonically decreasing energy*. B '07, Burke-Ortner-Süli '10, '13.
  - Backtracking algorithm: additional optimality conditions in trajectory space.
- mef90: unstructured parallel finite element, based on MPI + PETSc, scales up to thousands of cores. <http://bitbucket.org/bourdin/mef90-sieve>  
docker hub: bourdin/mef90-mpich
- *Coupled* minimization algorithms (Farrell-Maurini *IJNME* '17, Gerasimov-De Lorenzis, *CMAME* '17, Wick ): fast in convex regions (stable crack propagation), unstable / unpredictable otherwise.

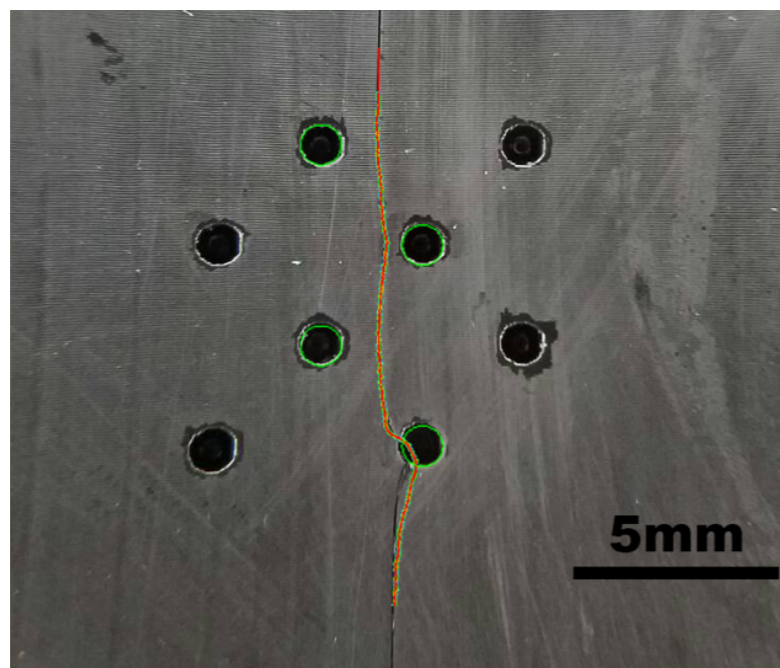
# Variational Phase Field models



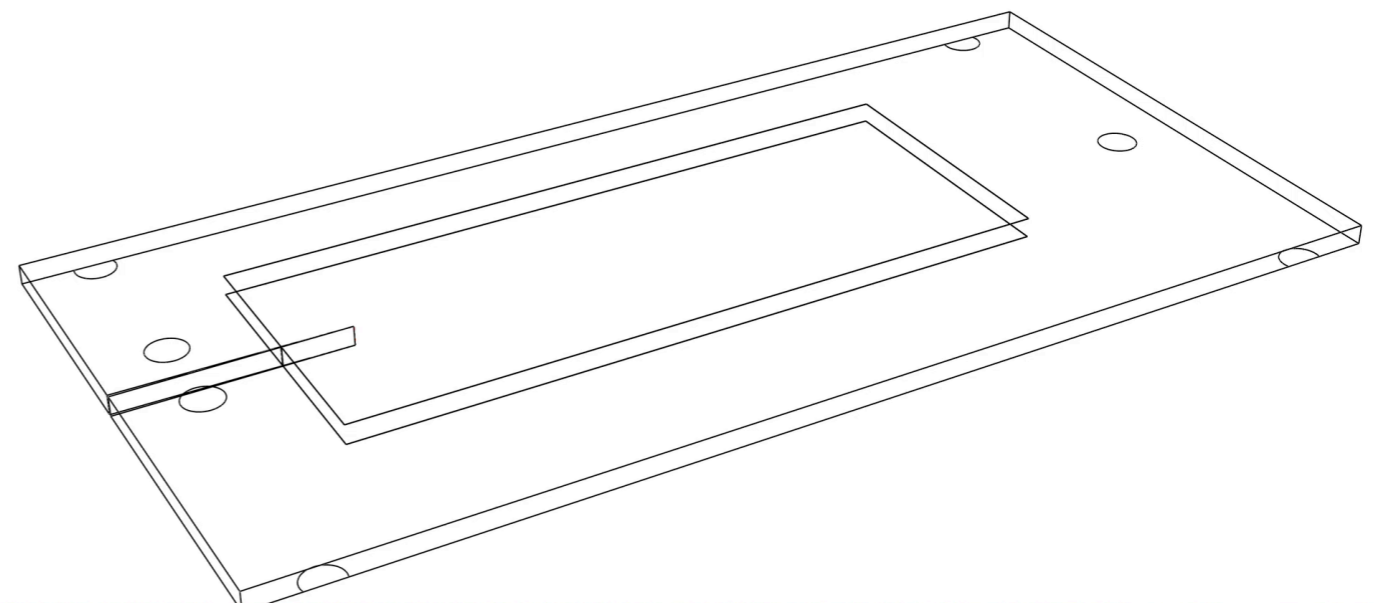
Leon-Baldelli et al *JMPS*, 2014



B-Chukwudozie-Yoshioka *SPE ATCE*, 2012



C.-J. Hsueh and N. Broadnik



# “our” phase-fields vs. “your” phase-field

Modica-Mortola (rescaled GL free energy): *double well* potential.

$$\mathcal{P}_\ell(\alpha) = \frac{1}{2c_W} \int_{\Omega} \frac{W(\alpha)}{\ell} + \ell |\nabla \alpha|^2 dx$$

- Free *boundary* problem, blow-up limits = surfaces with constant curvature

$$\mathcal{P}_\ell(\alpha) \xrightarrow{\Gamma} \mathcal{H}^{N-1}(\partial D), \quad \alpha \rightarrow \chi_D;$$

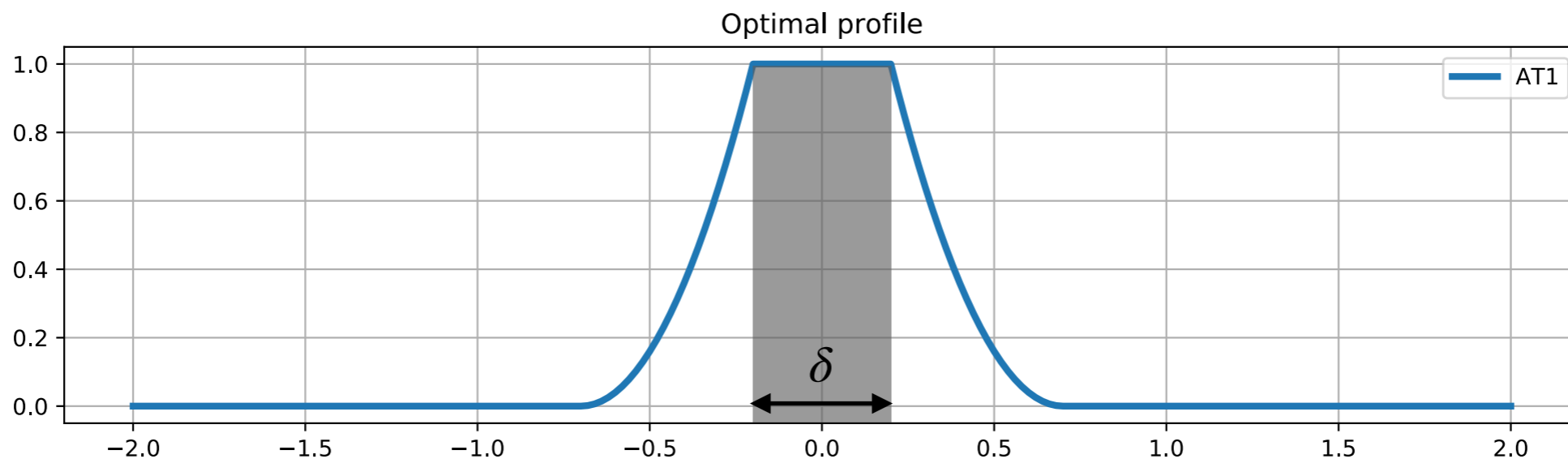
- Gradient-flow converges to mean-curvature motion;
- Cannot *localize* (nucleation requires concentrated driving forces);
- Interfaces can (*should*) move even without driving force.

# “our” phase-fields vs. “your” phase-field

Ambrosio-Tortorelli: *single well* potential.

$$\mathcal{E}_\ell(u, \alpha) := \int_{\Omega} a(\alpha) W(e(u)) dx + \frac{G_c}{4c_w} \int_{\Omega} \frac{w(\alpha)}{\ell} + \ell |\nabla \alpha|^2 dx$$

- Free *discontinuity* problem, blow up limits (2D): line, Y junction, crack tip;
- Evolution by unilateral minimization (criticality + stability). Convergence of gradient flows is an open problem;
- Interfaces cannot move under non-singular driving forces (irreversibility + energy balance), crack-tip motion requires driving force in  $\mathcal{O}(1/r)$ ;



Surface energy:  $+\mathcal{O}(\delta/\ell)$

Elastic energy:  $-\mathcal{O}(\delta)$

- Nucleation possible even with constant driving force.

# AT1 Surfing

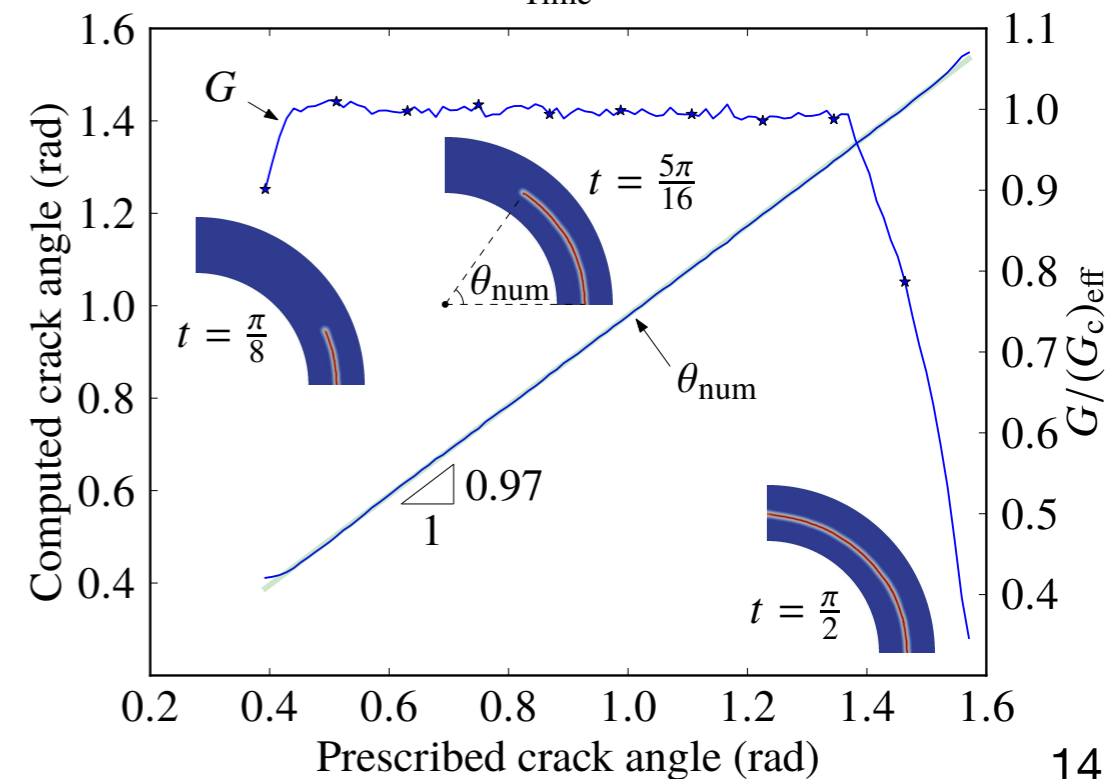
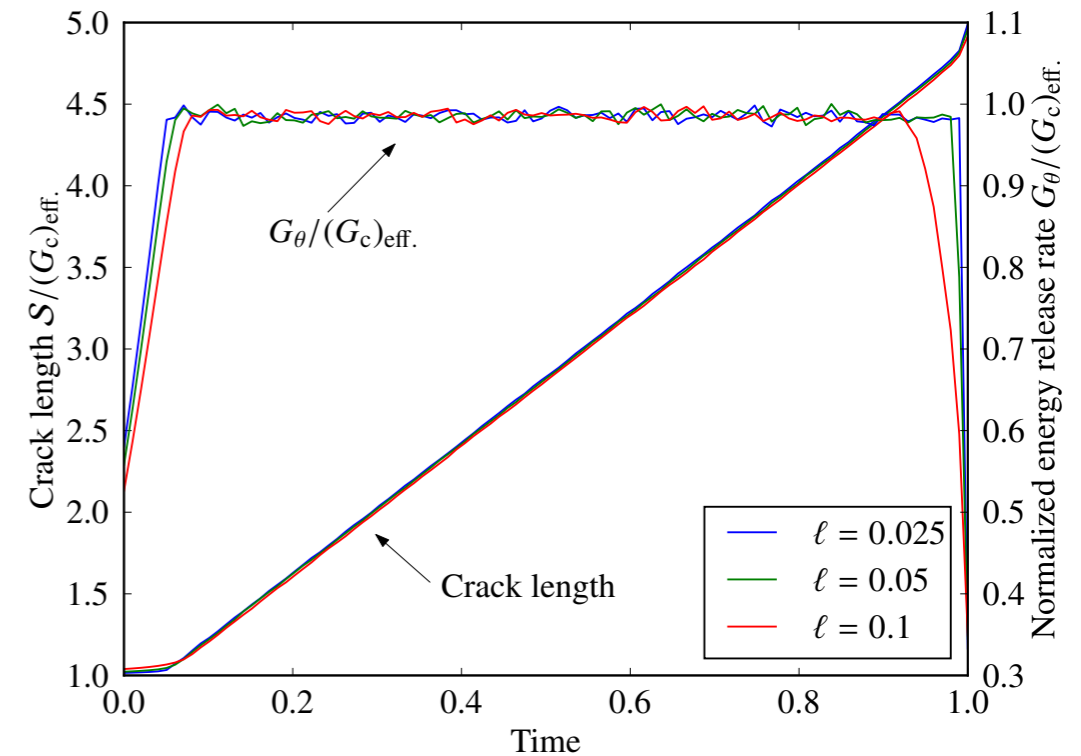
Homogeneous isotropic material with initial crack, AT1 model.

Elastic energy release rate computed with the  $G_\theta$  method or J-integral

Translating boundary displacement.

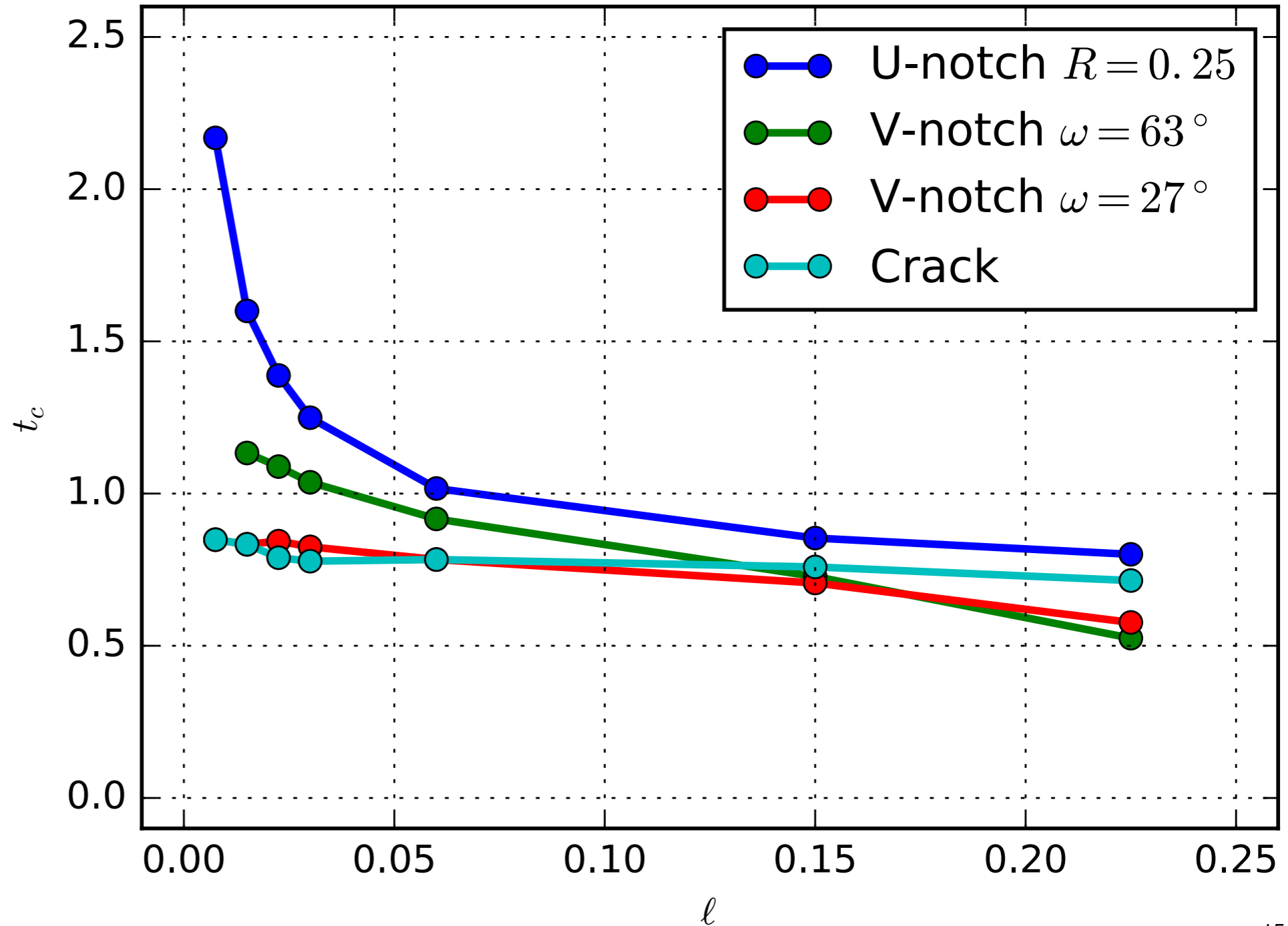
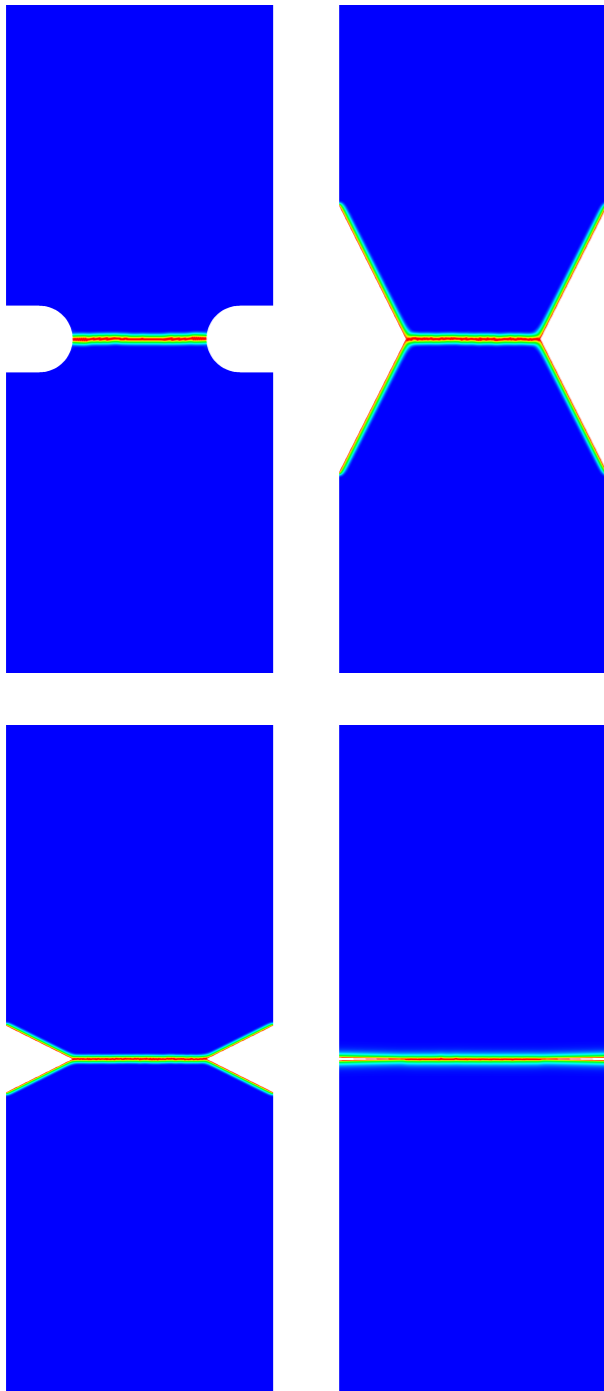


AT1 model using a fixed ratio  $\ell/h = 5$

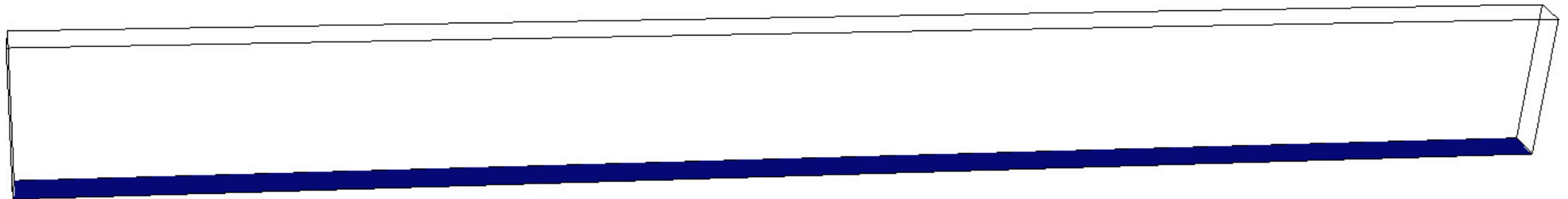
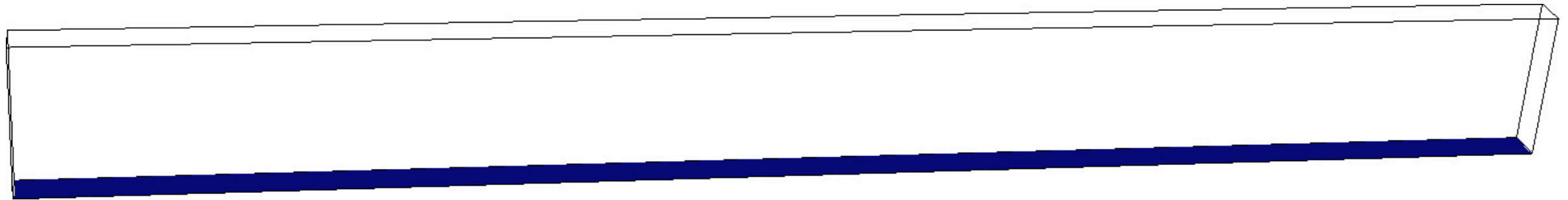


# Nucleation

- Double Edge Notch test with 4 notch geometries, uniaxial tension
- Fixed geometry, nucleation load vs  $\ell$ ,  $h/\ell$  constant

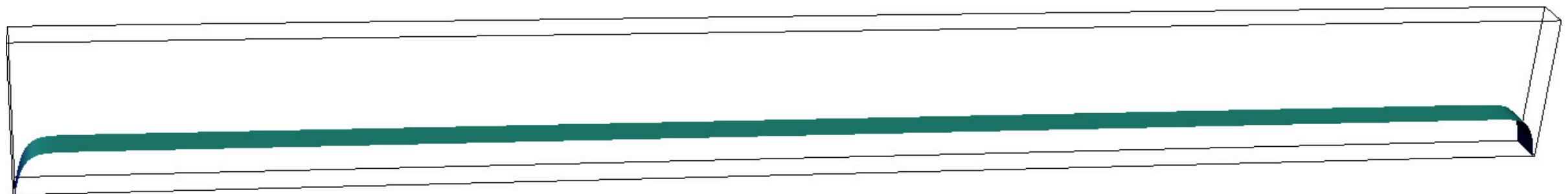
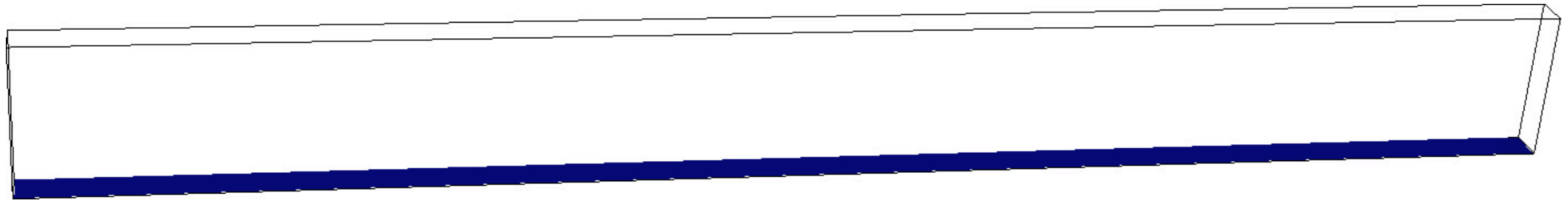


# 1D problem - AT1 vs AT2





# 1D problem - AT1 vs AT2



# 1D problem

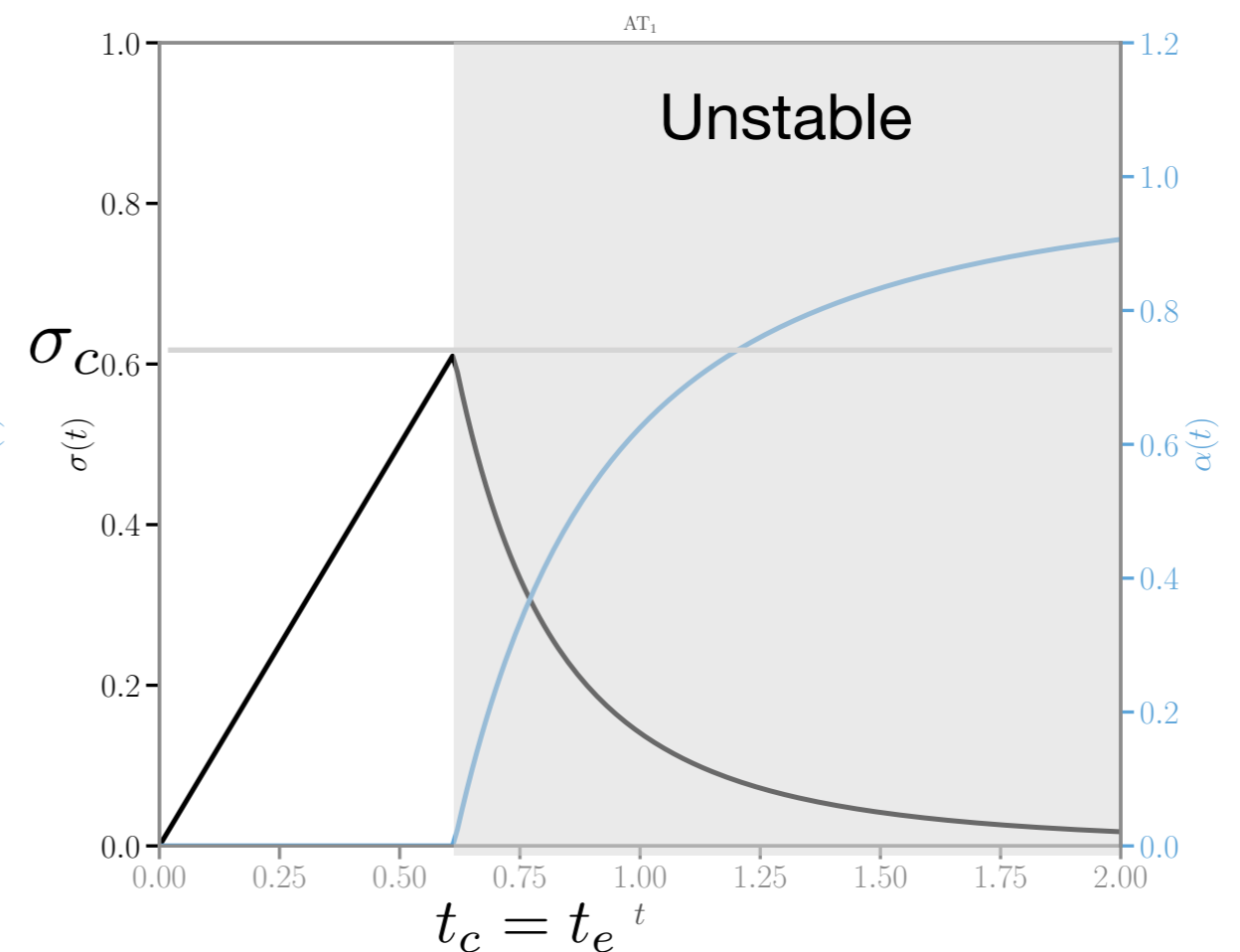
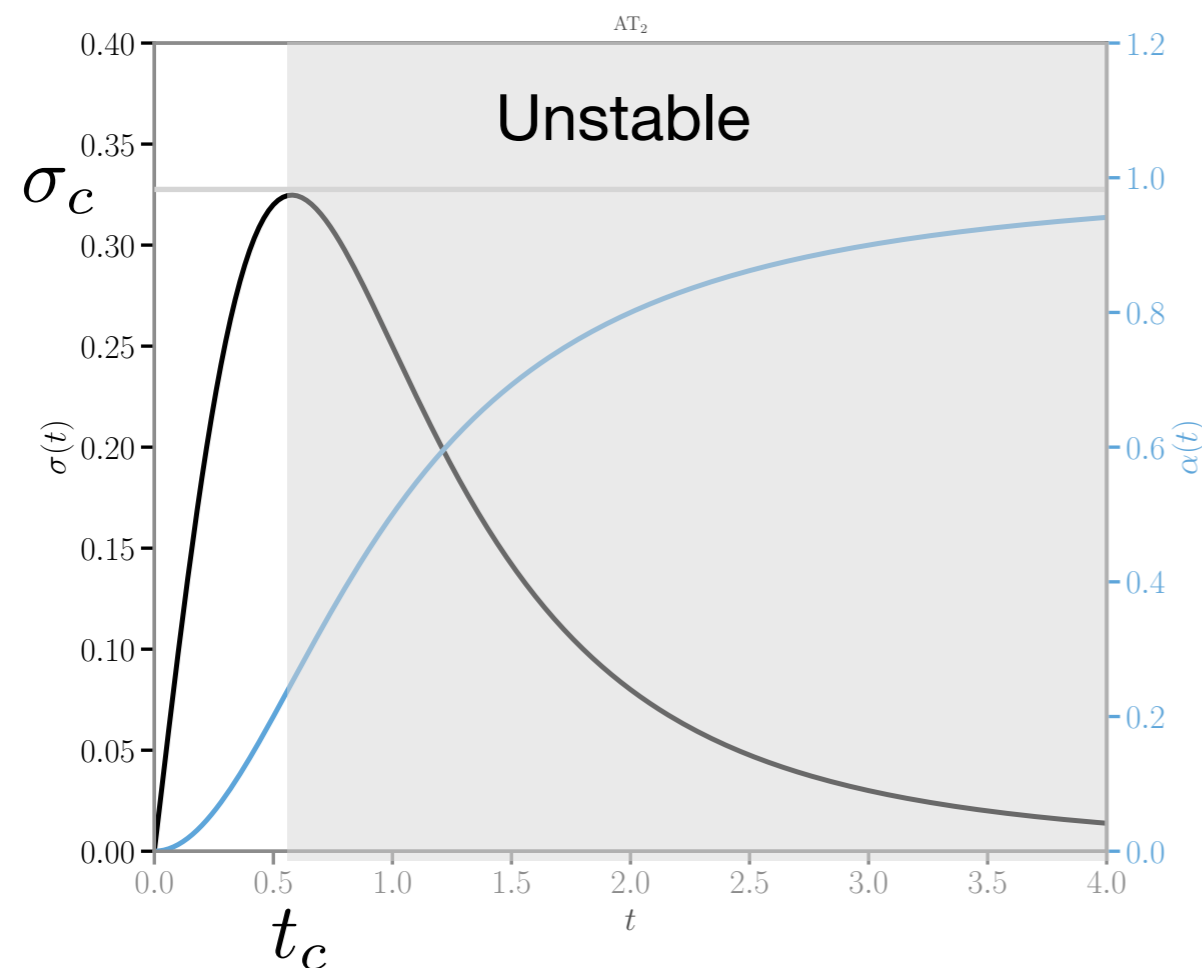


AT1 and AT2 models behave differently at *fixed*  $\ell$ :

$$\text{AT2: } \mathcal{E}_\ell(u, \alpha) = \int_{\Omega} (1 - \alpha)^2 W(e(u)) dx + \frac{G_c}{2} \int_{\Omega} \frac{\alpha^2}{\ell} + \ell |\nabla \alpha|^2 dx$$

$$\text{AT1: } \mathcal{E}_\ell(u, \alpha) = \int_{\Omega} (1 - \alpha)^2 W(e(u)) dx + \frac{3G_c}{8} \int_{\Omega} \frac{\alpha}{\ell} + \ell |\nabla \alpha|^2 dx$$

Construction of elastic and homogeneous states ( $\alpha = \text{cst}$ )



# 1D problem

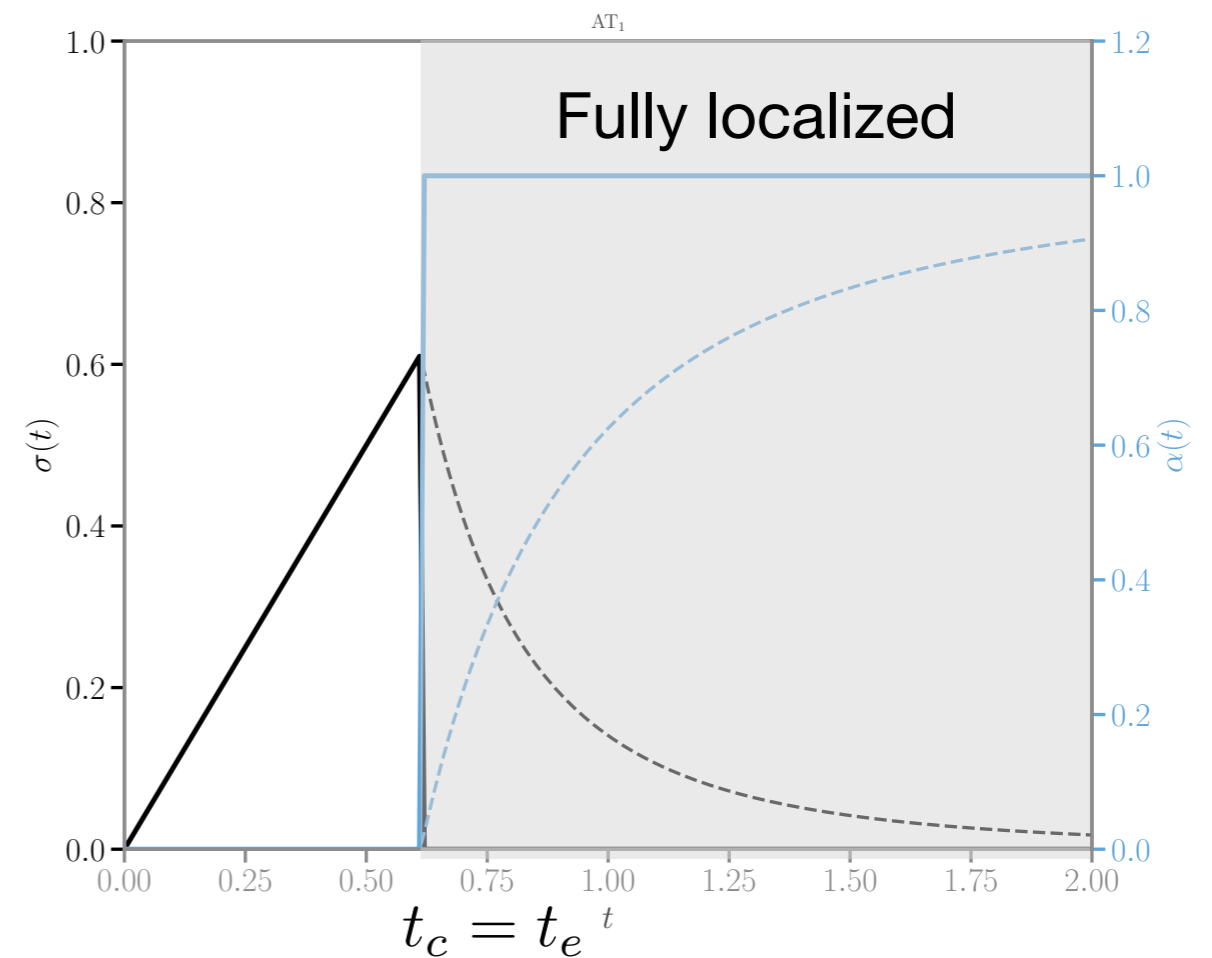
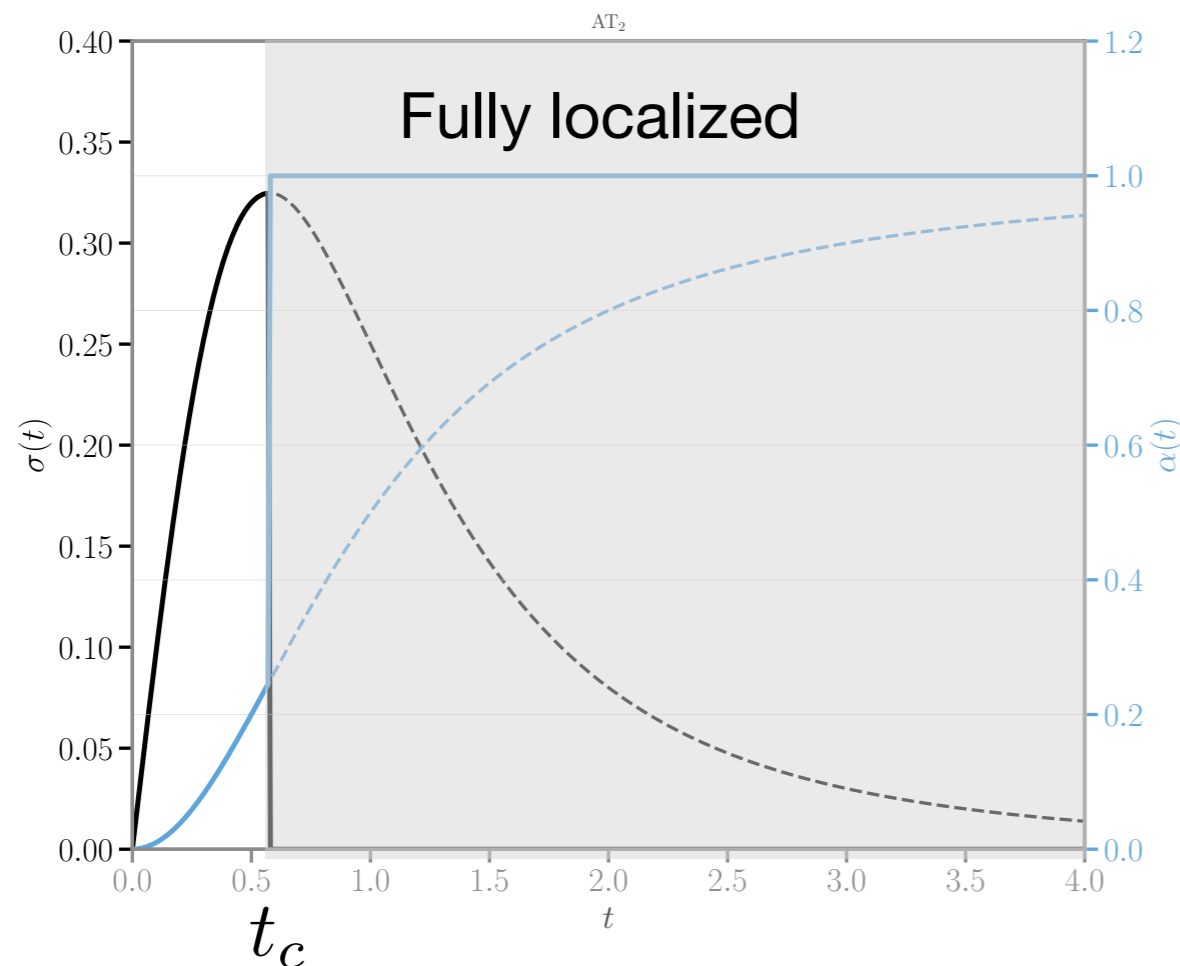


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$$\text{AT2: } \mathcal{E}_\ell(u, \alpha) = \int_{\Omega} (1 - \alpha)^2 W(e(u)) dx + \frac{G_c}{2} \int_{\Omega} \frac{\alpha^2}{\ell} + \ell |\nabla \alpha|^2 dx$$

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Construction of elastic, homogeneous and localized states, stability analysis



# 1D problem



AT1 and AT2 models:

| model           | $w(\alpha)$ | $a(\alpha)$      | $\sigma_e$                    | $\sigma_c$                                | $D$      | $c_w$         |
|-----------------|-------------|------------------|-------------------------------|---|----------|---------------|
| AT <sub>1</sub> | $\alpha$    | $(1 - \alpha)^2$ | $\sqrt{\frac{3G_c E}{8\ell}}$ | $\sqrt{\frac{3G_c E}{8\ell}}$             | $4\ell$  | $\frac{2}{3}$ |
| AT <sub>2</sub> | $\alpha^2$  | $(1 - \alpha)^2$ | 0                             | $\frac{3}{16} \sqrt{\frac{3G_c E}{\ell}}$ | $\infty$ | $\frac{1}{2}$ |

Regularization parameter  $\ell$  is *not* an independent parameter.

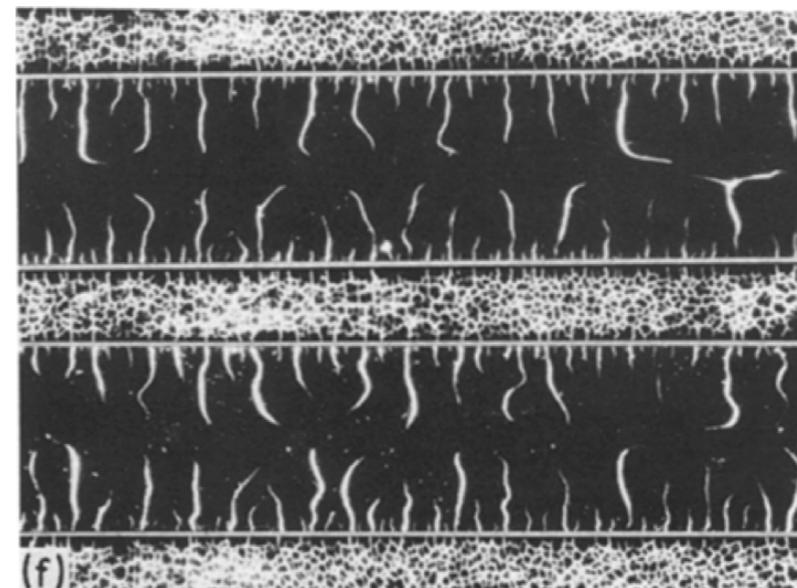
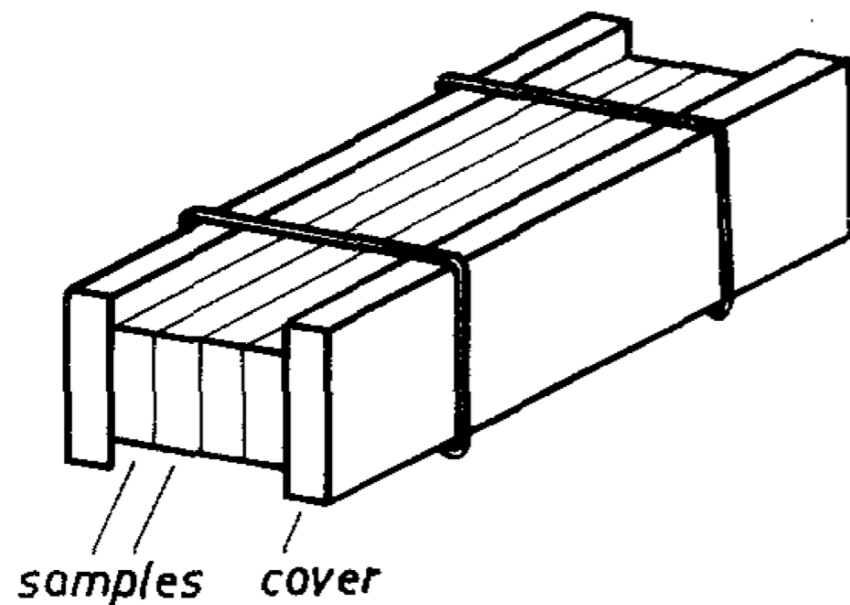
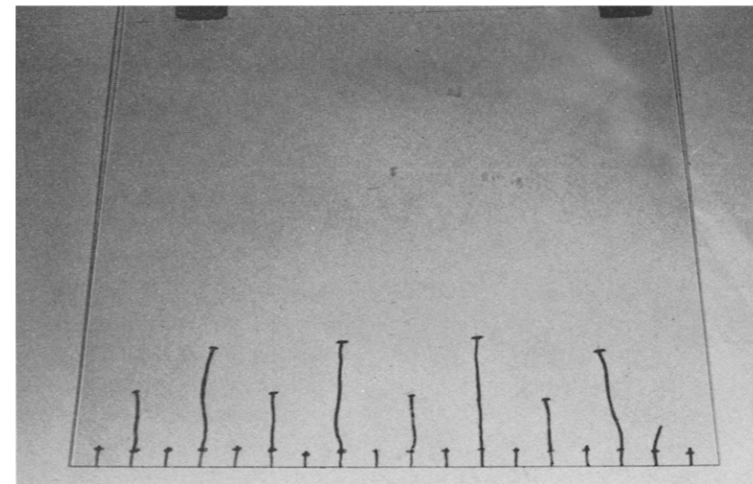
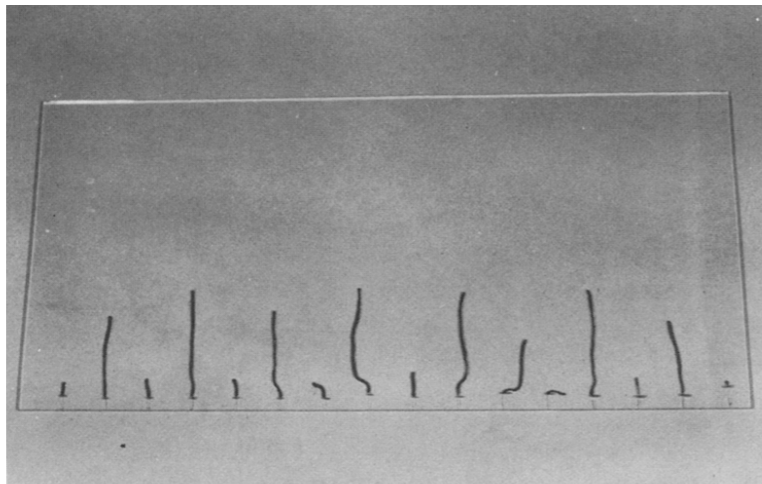
$\ell$  is the material's *characteristic/internal length*, depends on  $\sigma_c$ , and  $K_{Ic}$ :

$$\text{AT1: } \ell_1 := \frac{3 G_c E}{8 \sigma_c^2} = \frac{3 K_{Ic}^2}{8 \sigma_c^2}$$

$$\text{AT2: } \ell_2 := \frac{27 G_c E}{256 \sigma_c^2} = \frac{27 K_{Ic}^2}{256 \sigma_c^2}$$

# Thermal shock

- Lab experiment: thermal shock in glass, quenching of ceramic plates.
- Original motivation: thermal reservoir stimulation.
- Uncoupled problem: effect of crack geometry on heat transfer neglected.
- Dimensional analysis: only one parameter  $l_0 = \frac{G_c}{E(\alpha\Delta T)^2}$



# Thermal shock problem revisited

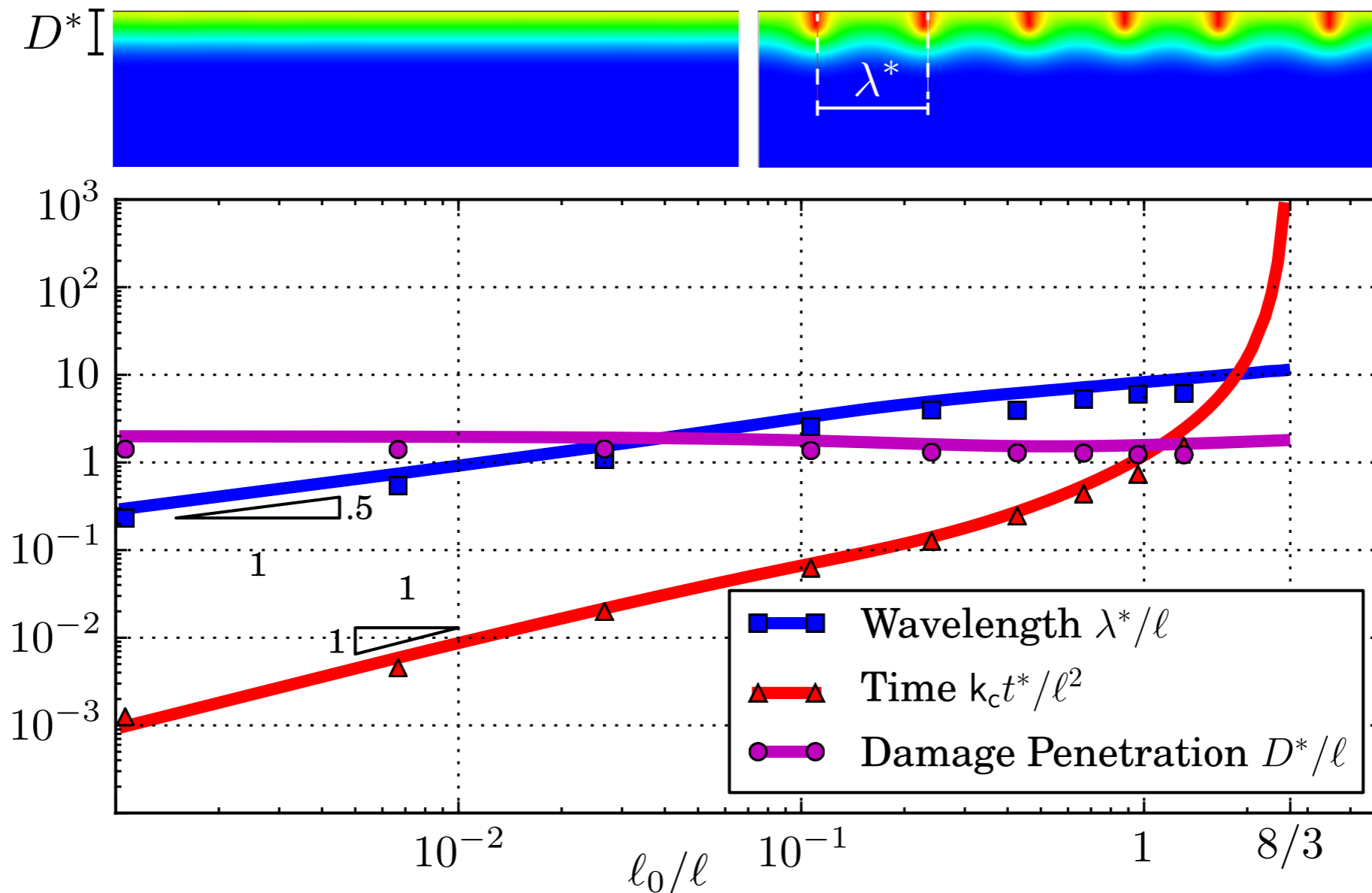
Construction of elastic and homogeneously damaged solutions

For high enough temperature contrast:

Elastic solution

Homogeneous damage

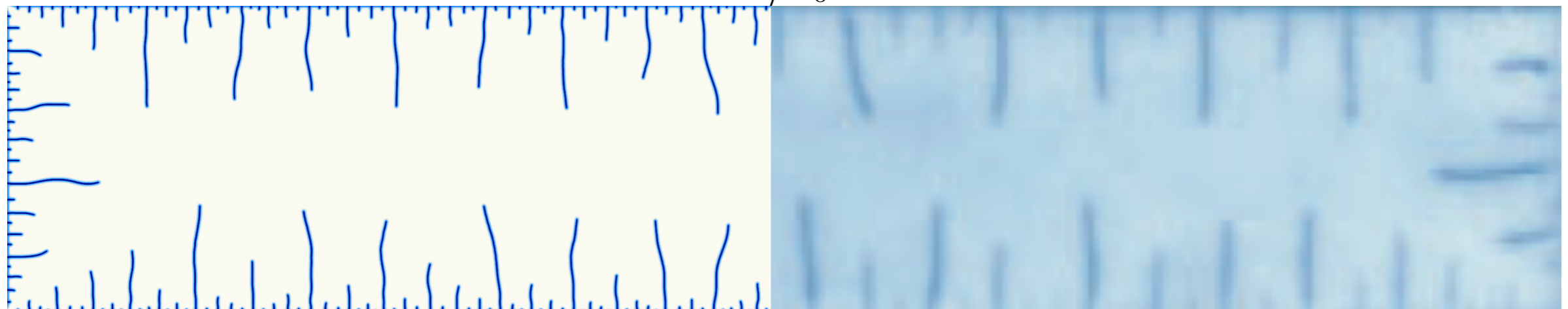
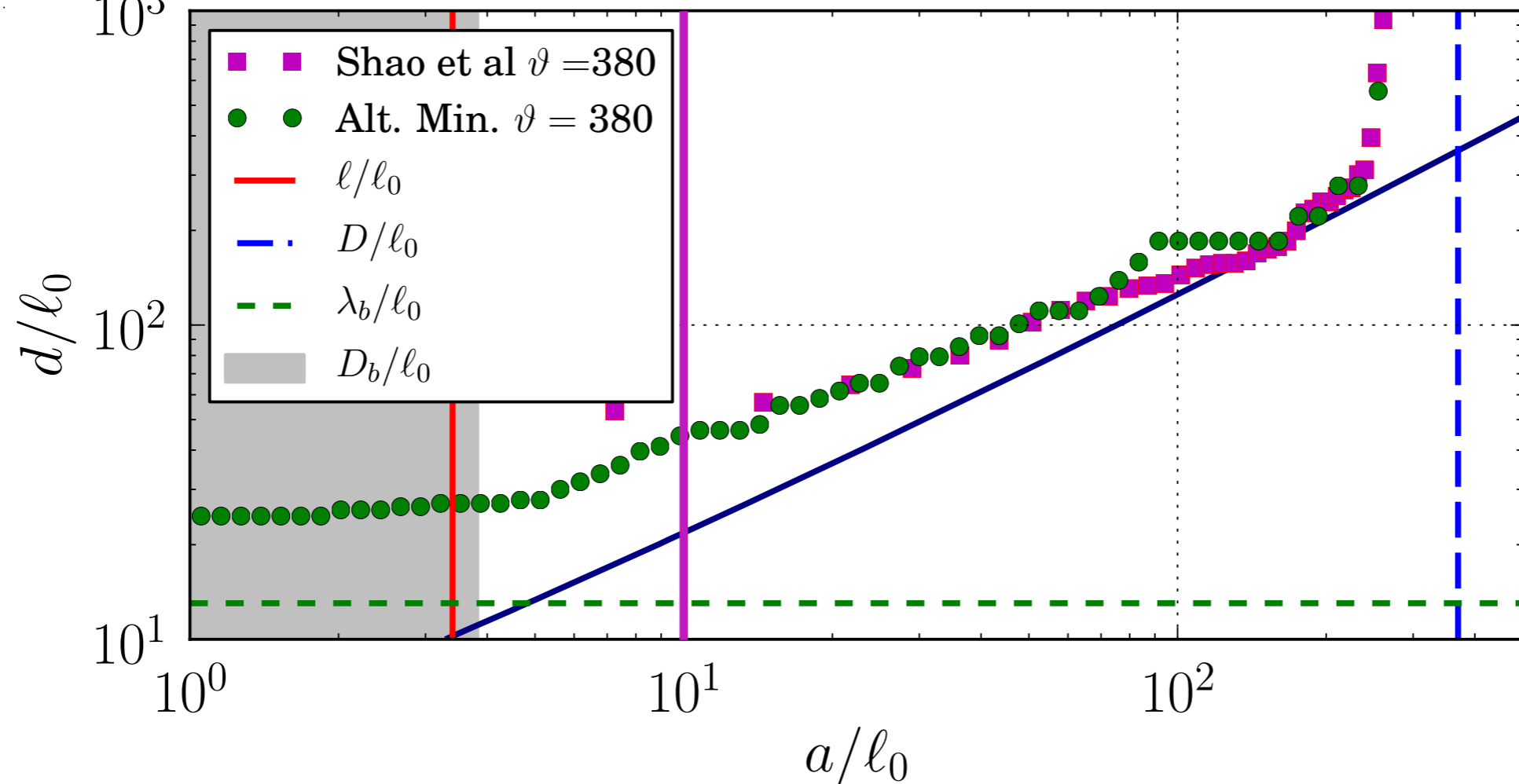
Localization with spacing  $\lambda$ , depth  $D$ .



# Validation: experiment vs. simulation

●  $E = 340 \text{ GPa}$ ,  $\nu = 0.22$ ,  $G_c = 42.47 \text{ Jm}^{-2}$ ,  $\sigma_c = 342.2 \text{ MPa}$ ,  $\ell = 456.24 \mu\text{m}$

●  $1 \times 10^{-5}$

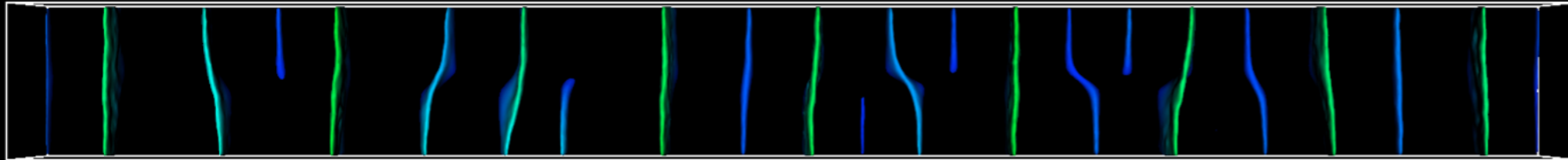


B-Marigo-Maurini-Sicsic, *PRL* 2014, Shao-Zhang-Xu-Zhou-Li-Liu, *J. AM. Ceram. Soc.* '11.

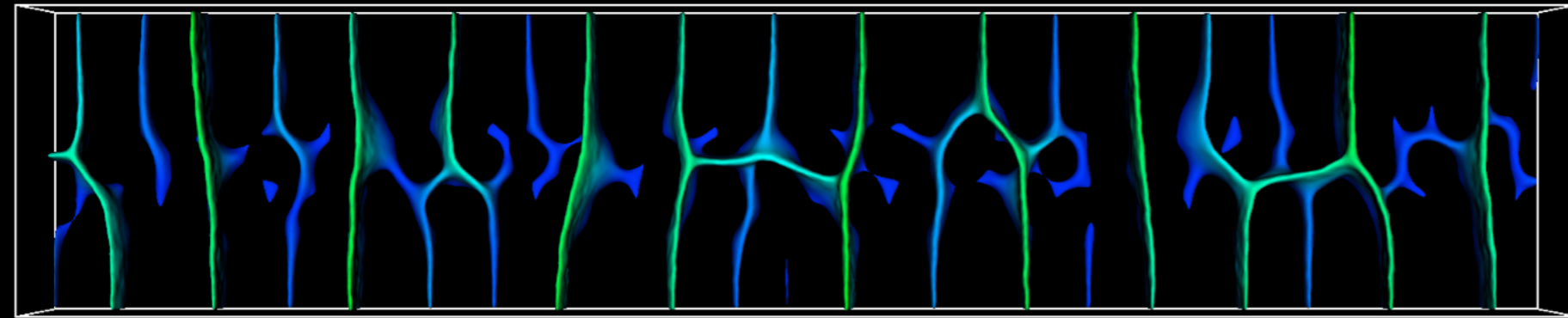
# 2d to 3d transition



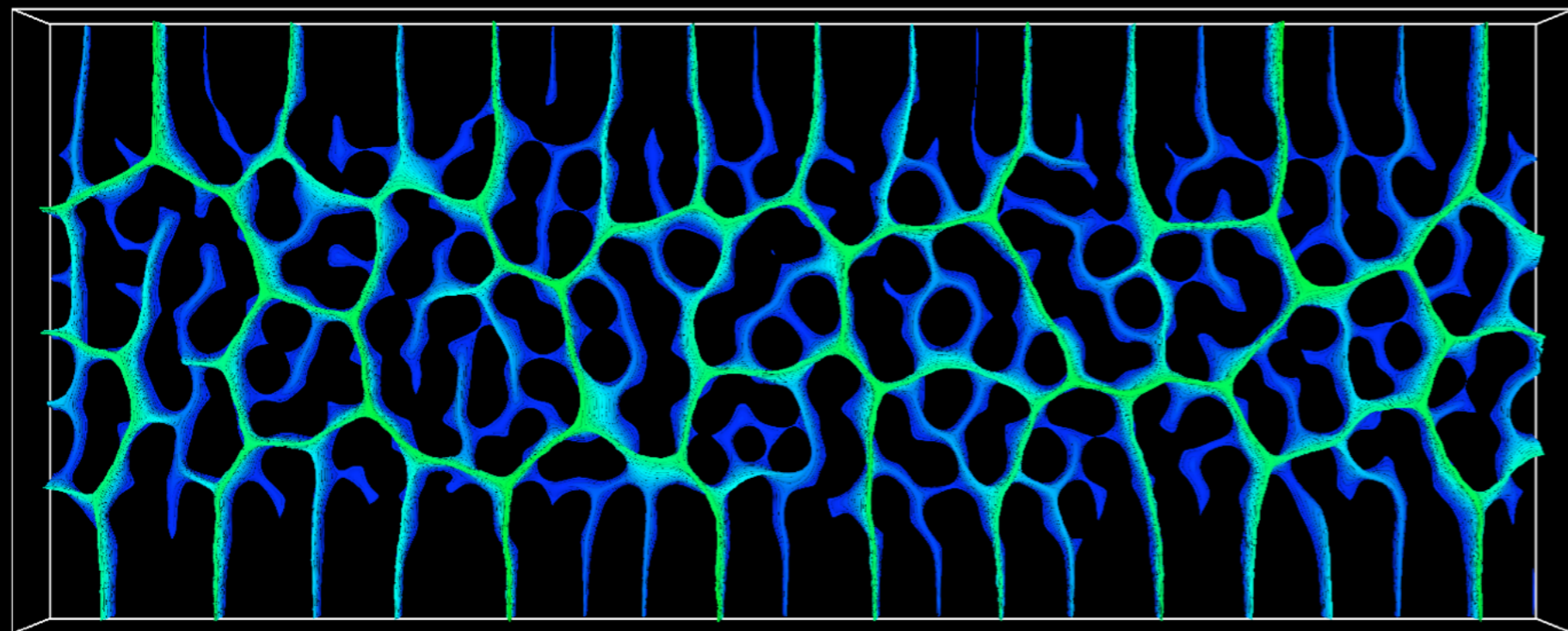
.1mm



.5mm



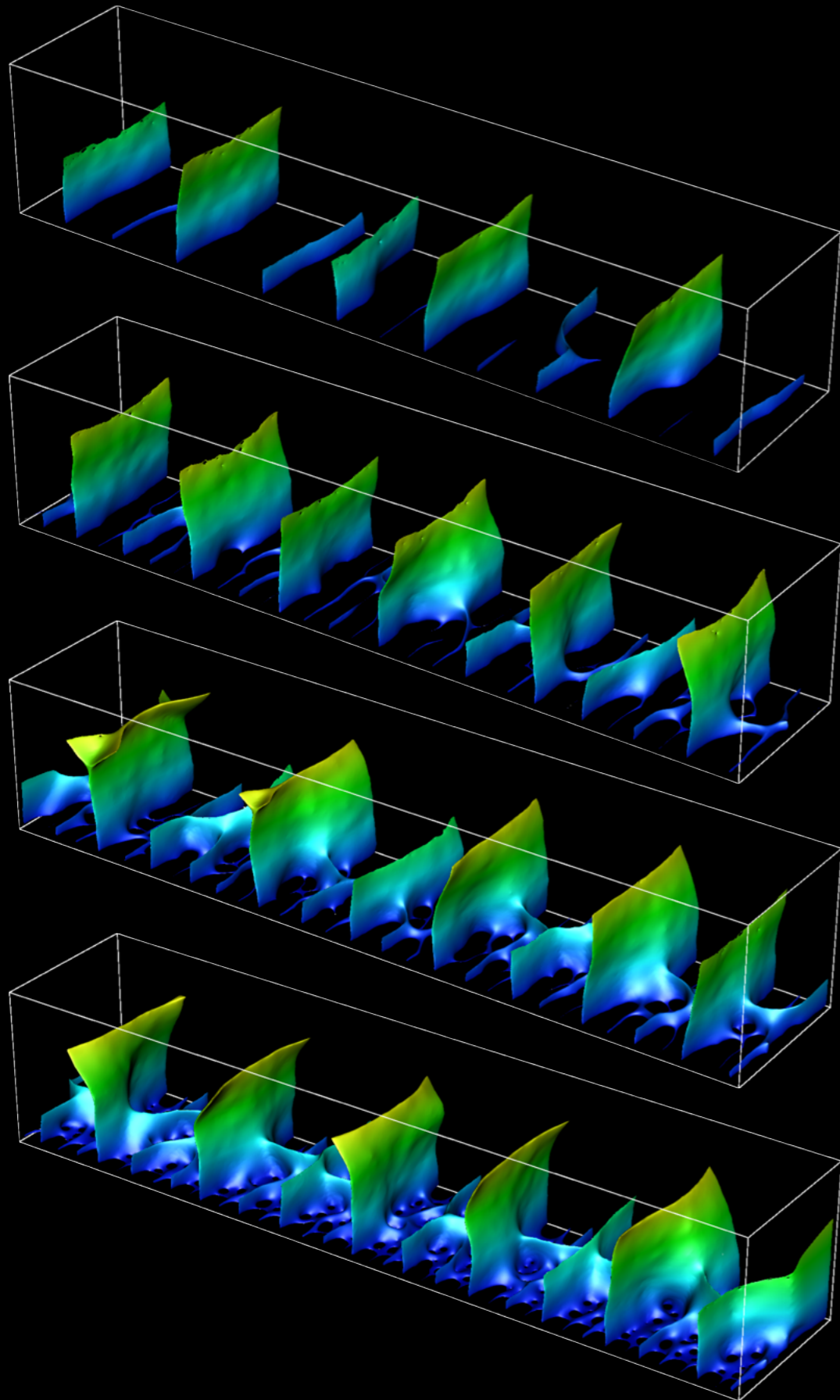
1mm



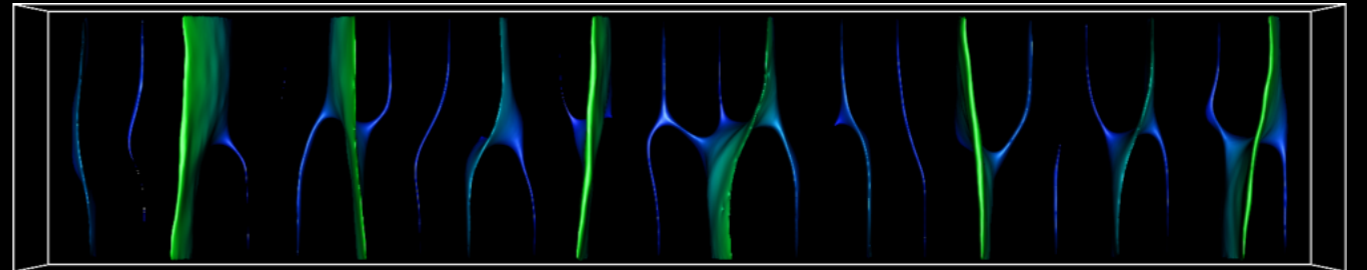
2mm



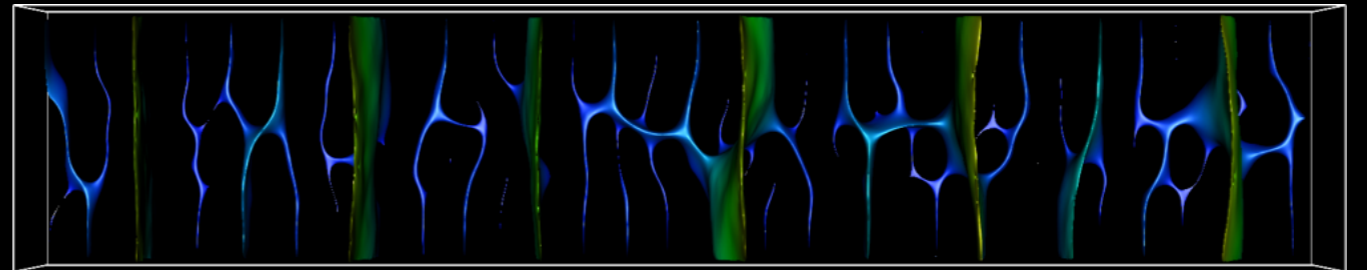
# 2d to 3d transition



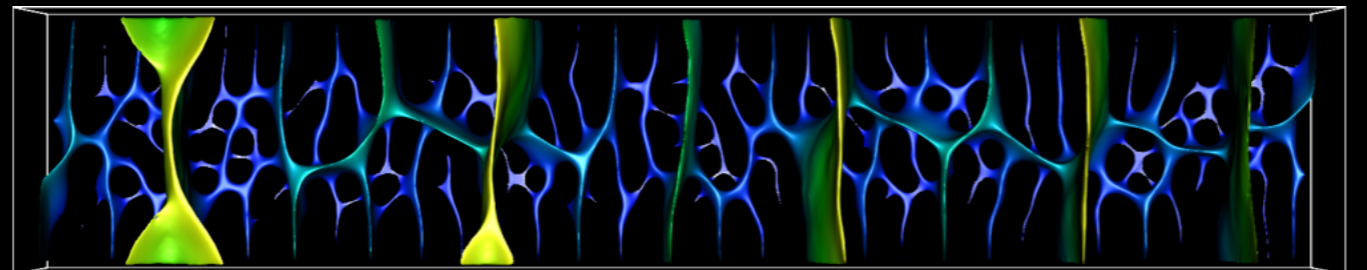
380K



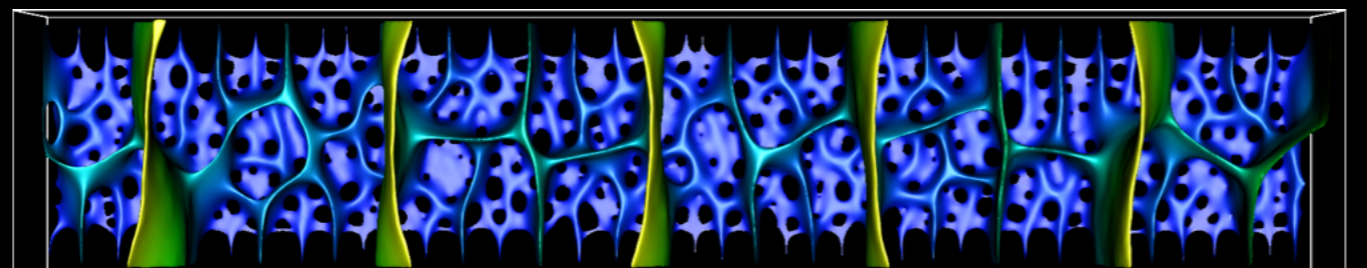
480K



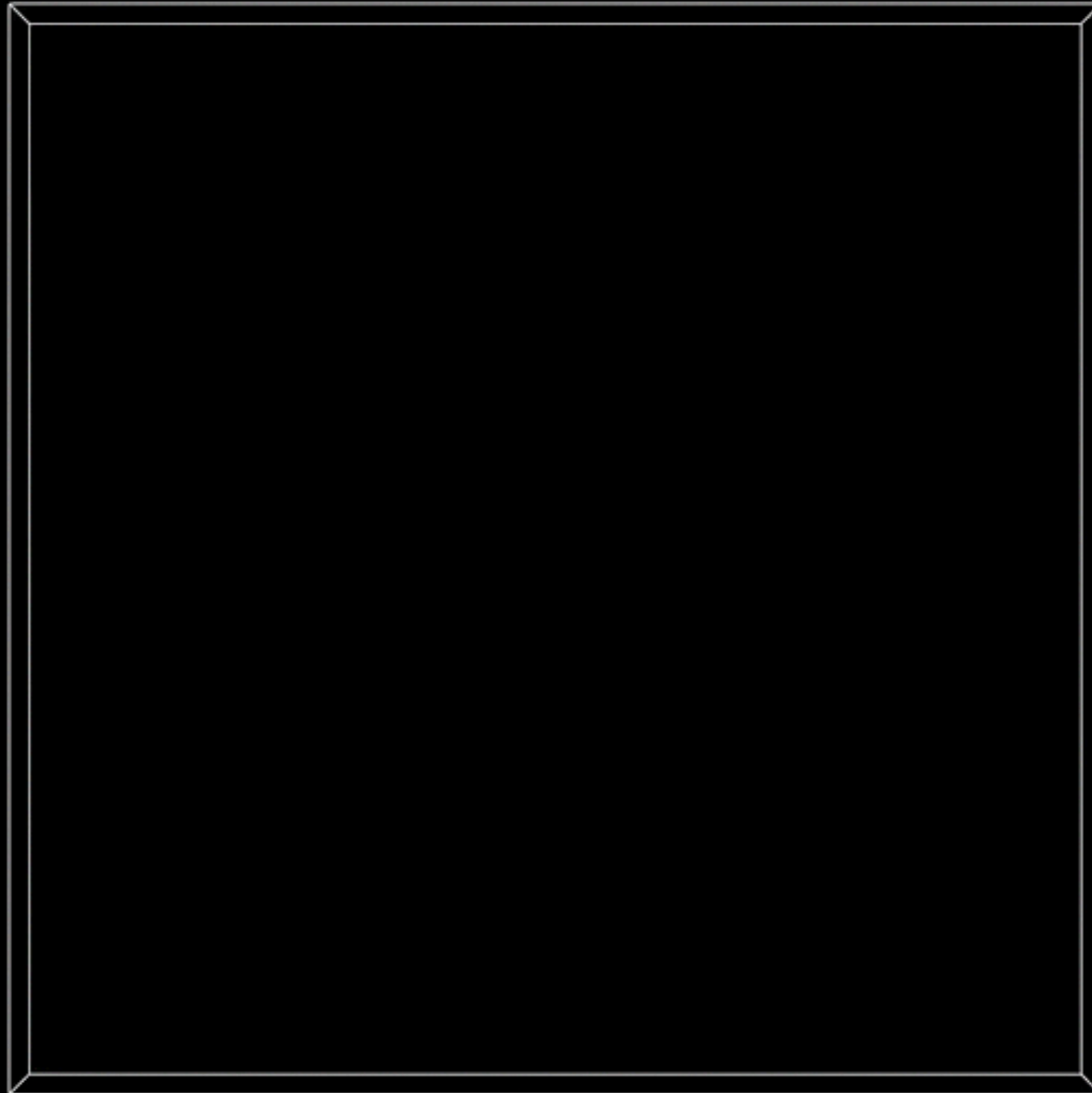
580K



680K

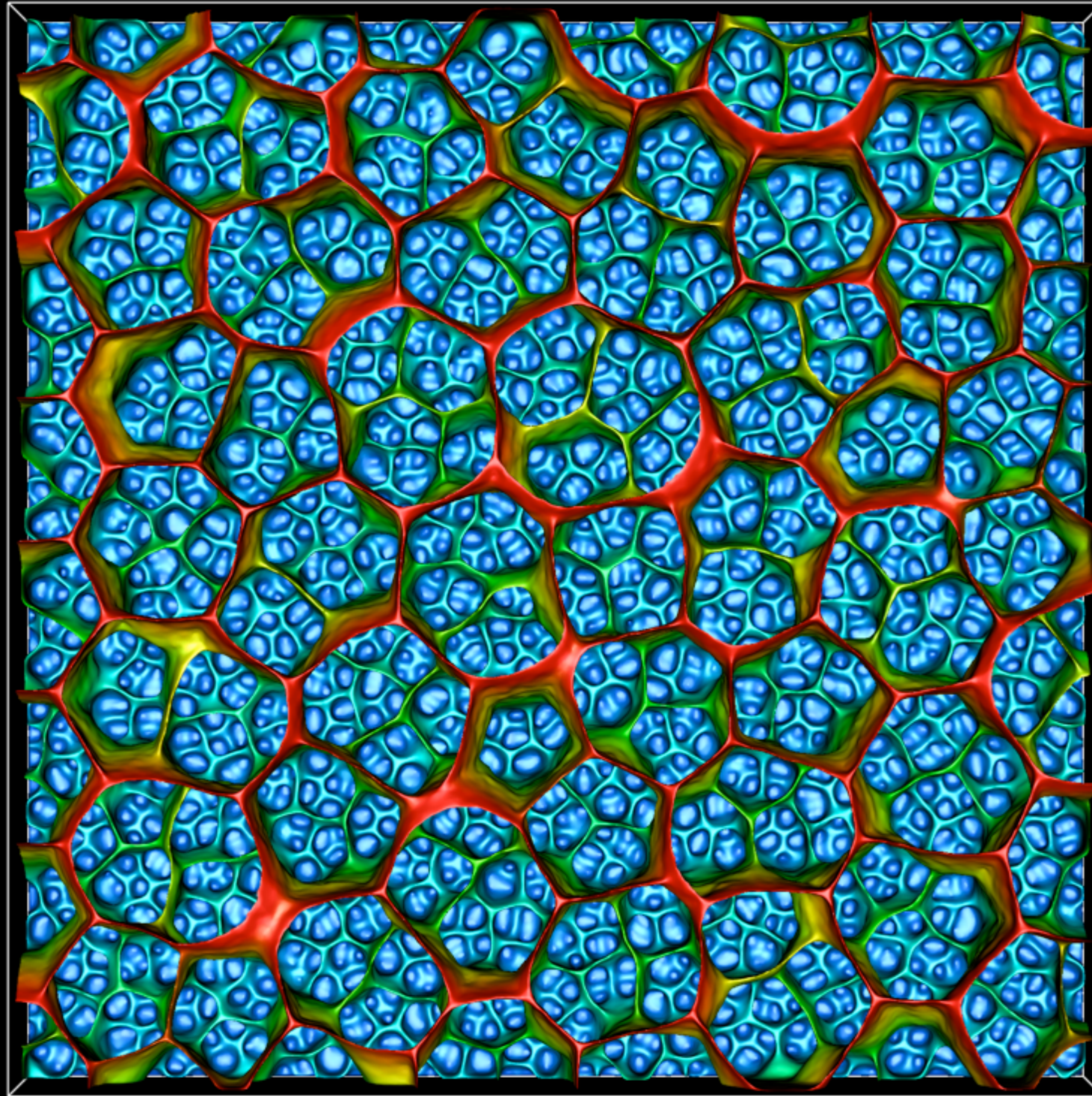


# Full 3d computation



44M elements, 1536 cores (stampede, TACC), 10h.

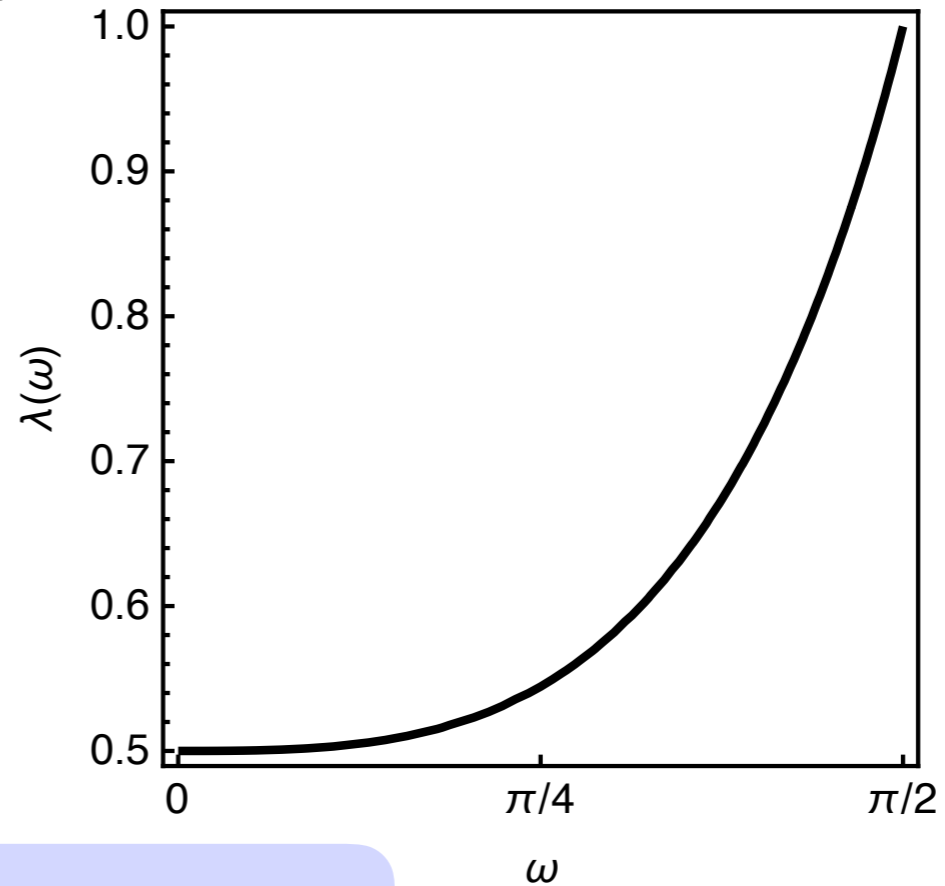
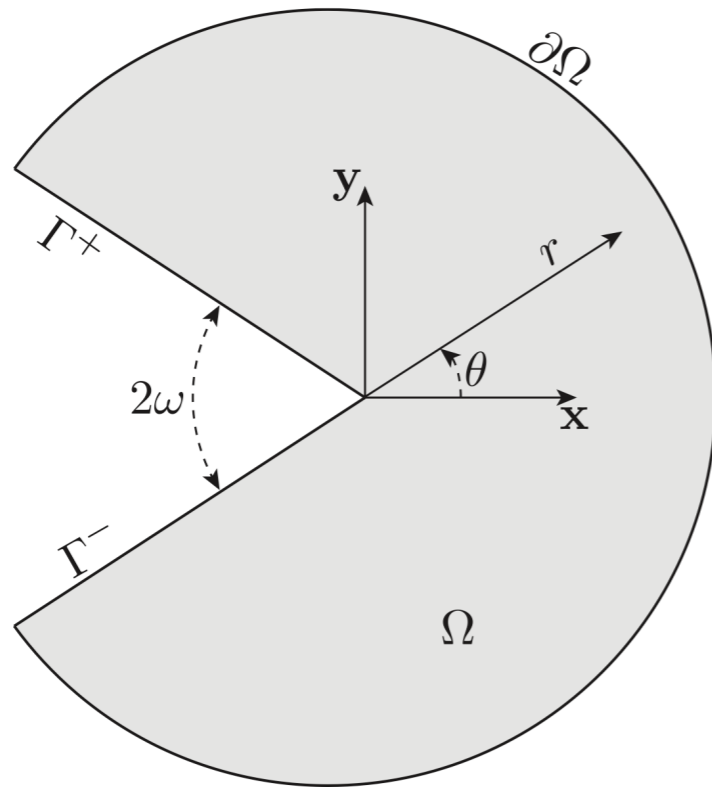
# Full 3d computation



44M elements, 1536 cores (stampede, TACC), 10h.

# Strength to toughness paradox

Notch angle  $2\omega$ , mode-I, singularity exponent  $\lambda(\omega)$ :



$$\sigma_{\theta\theta}(\theta = 0, r) = F(\theta)r^{\lambda-1}$$

Extreme cases:

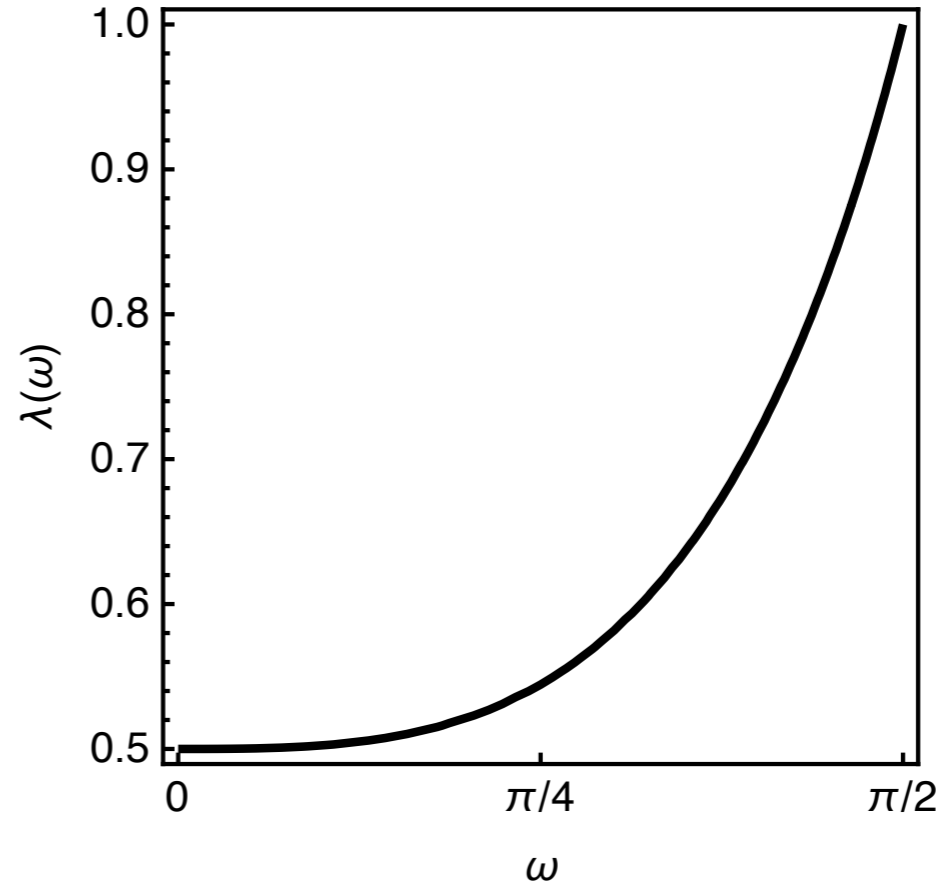
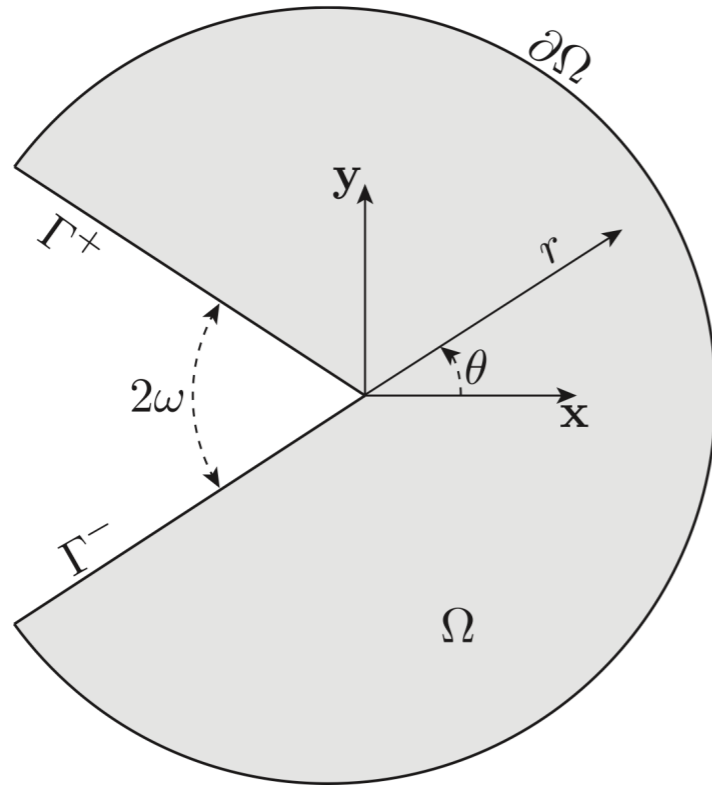
- $\omega = 0, \lambda=1/2$ : Griffith regime, progressive growth when  $K_I = K_{Ic}$ .
- $\omega \sim \pi/2, \lambda=1$ : uniaxial stress state near notch tip, nucleation at  $\sigma_c$ .

Intermediate angles:

- Nucleation impossible, according to Griffith / “local minimization”.
- Critical stress always exceeded (“weak” singularity).

# Initiation at a notch

Notch angle  $2\omega$ , mode-I, singularity exponent  $\lambda(\omega)$ :



Generalized Stress Intensity Factor:

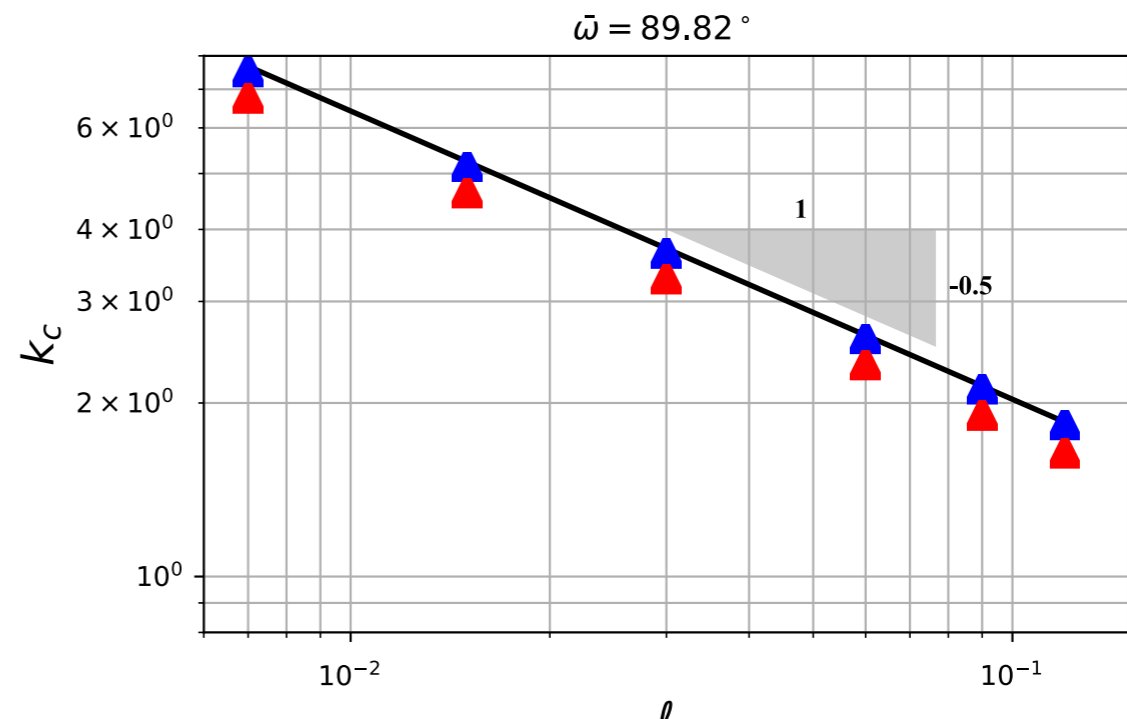
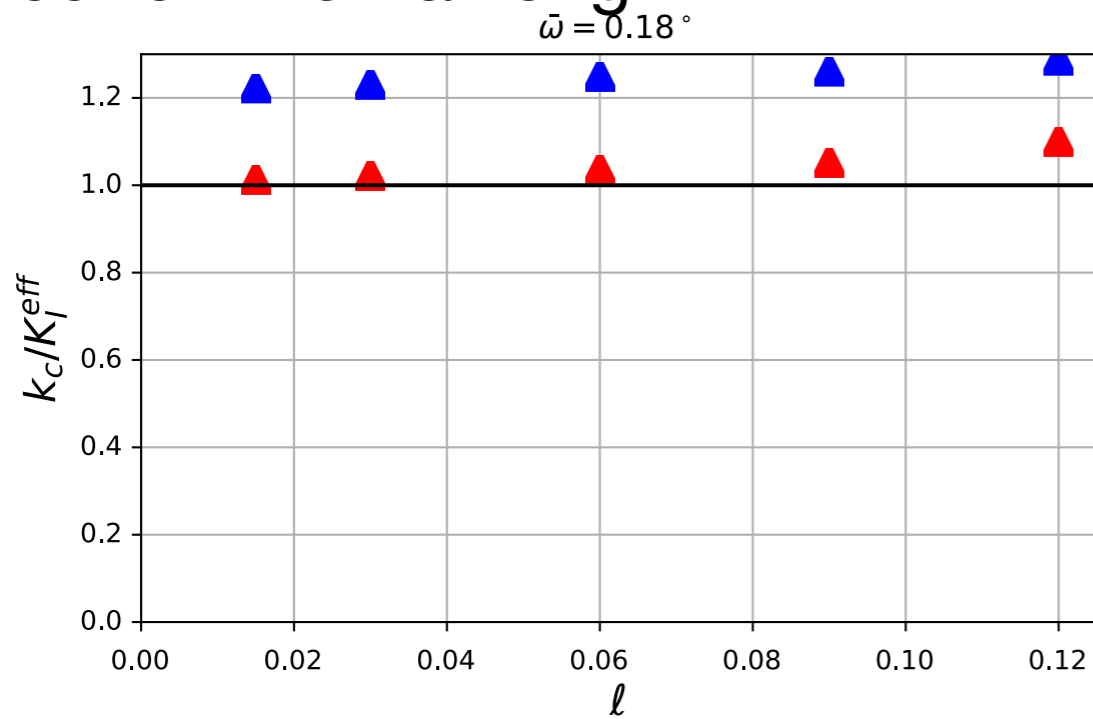
$$k = \lim_{r \rightarrow 0} \frac{\sigma_{\theta\theta}(r, 0)}{(2\pi r)^{\lambda-1}}$$

Extreme cases:

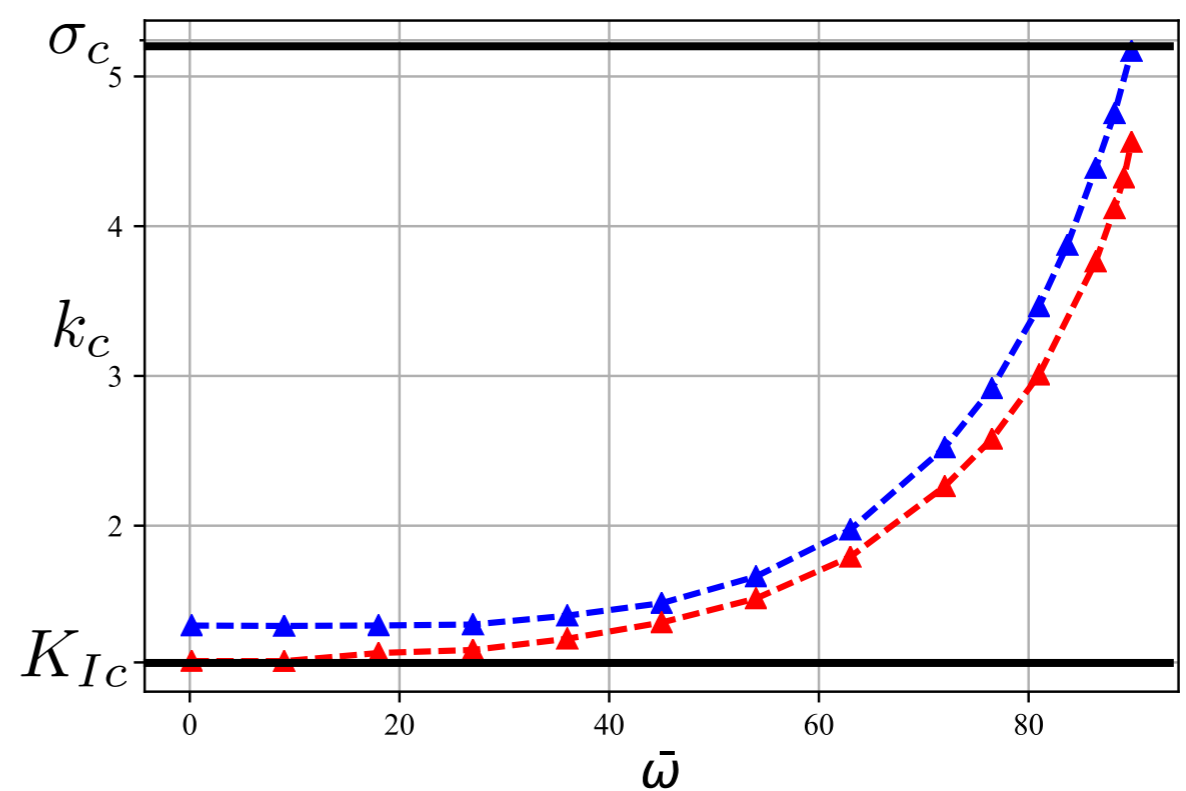
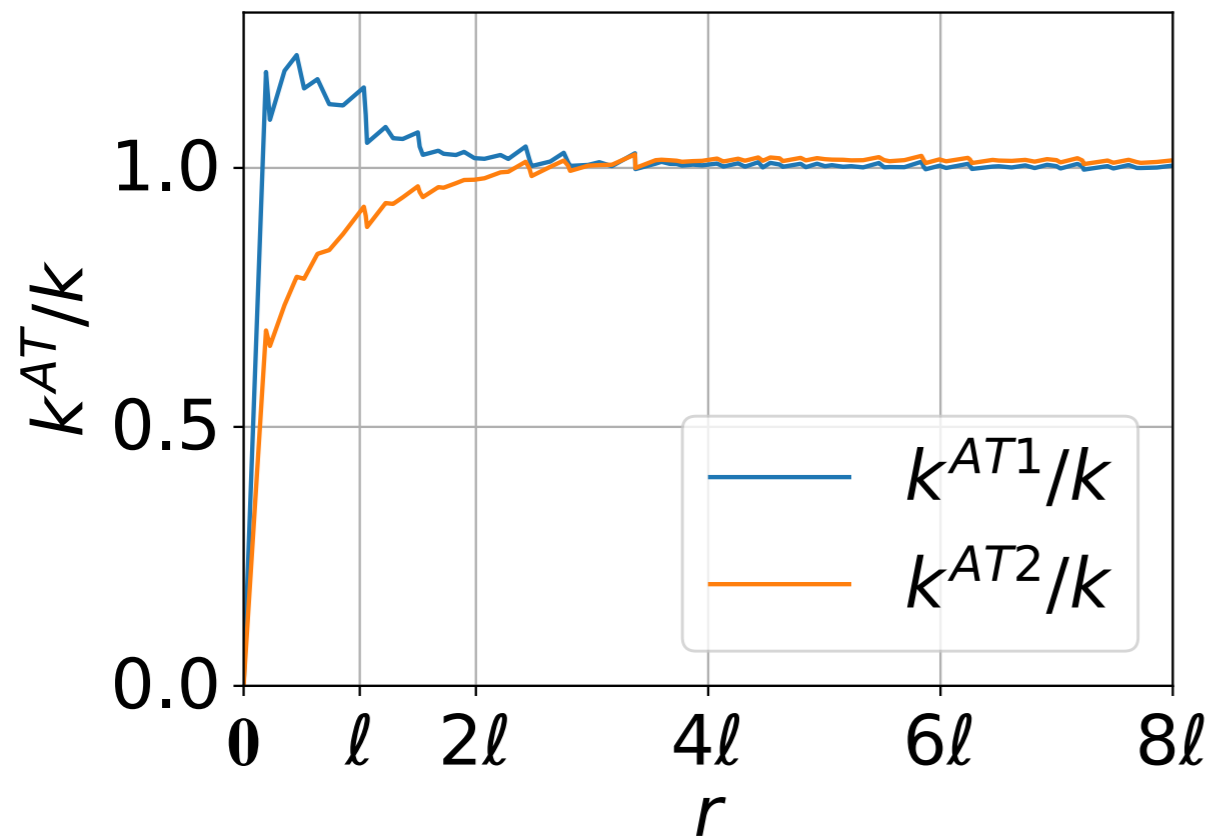
- $\omega = 0, k = K_I$ .
- $\omega = \pi/2, k = \sigma_{yy}$ .

# Initiation at a notch, AT1

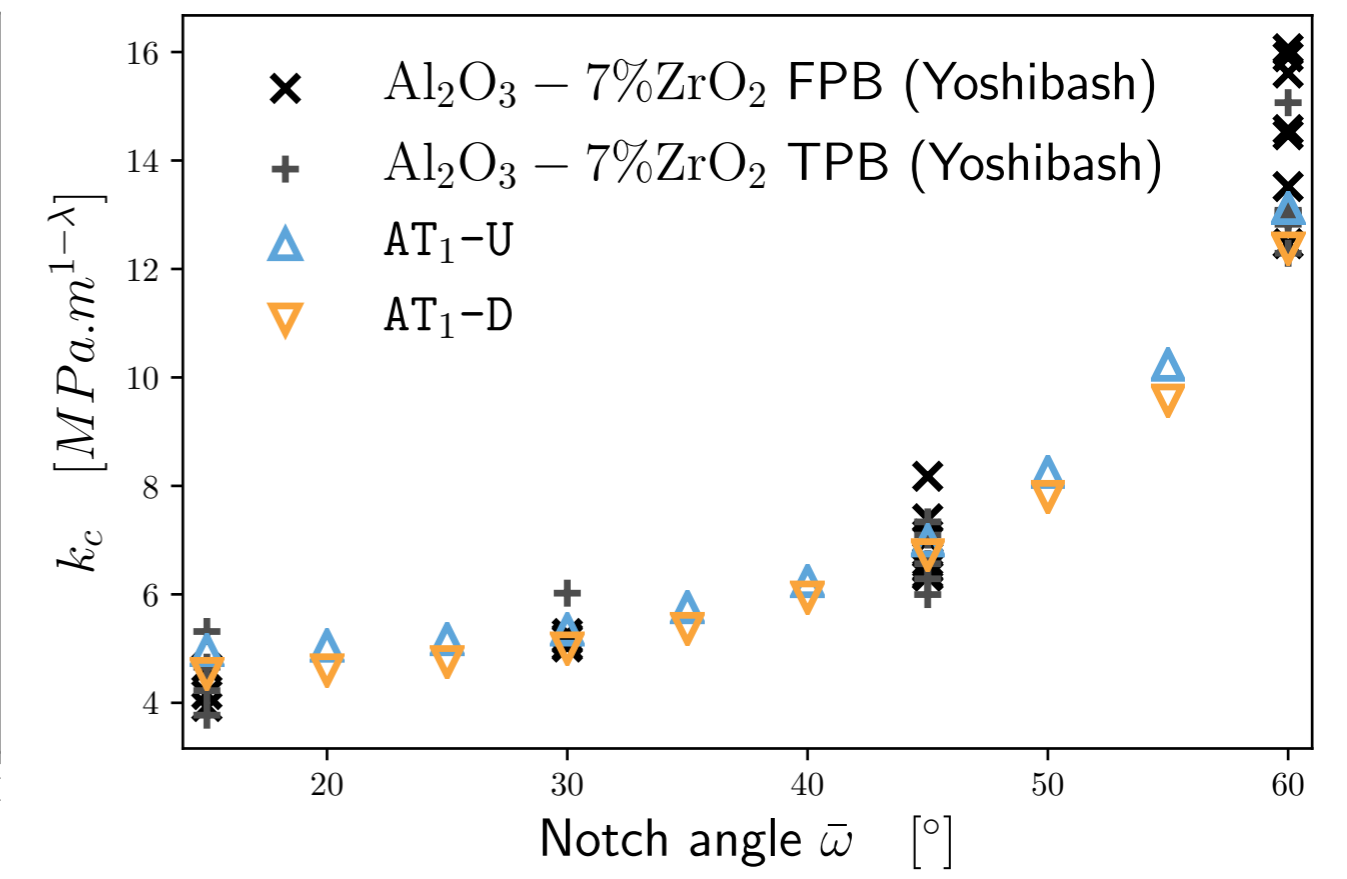
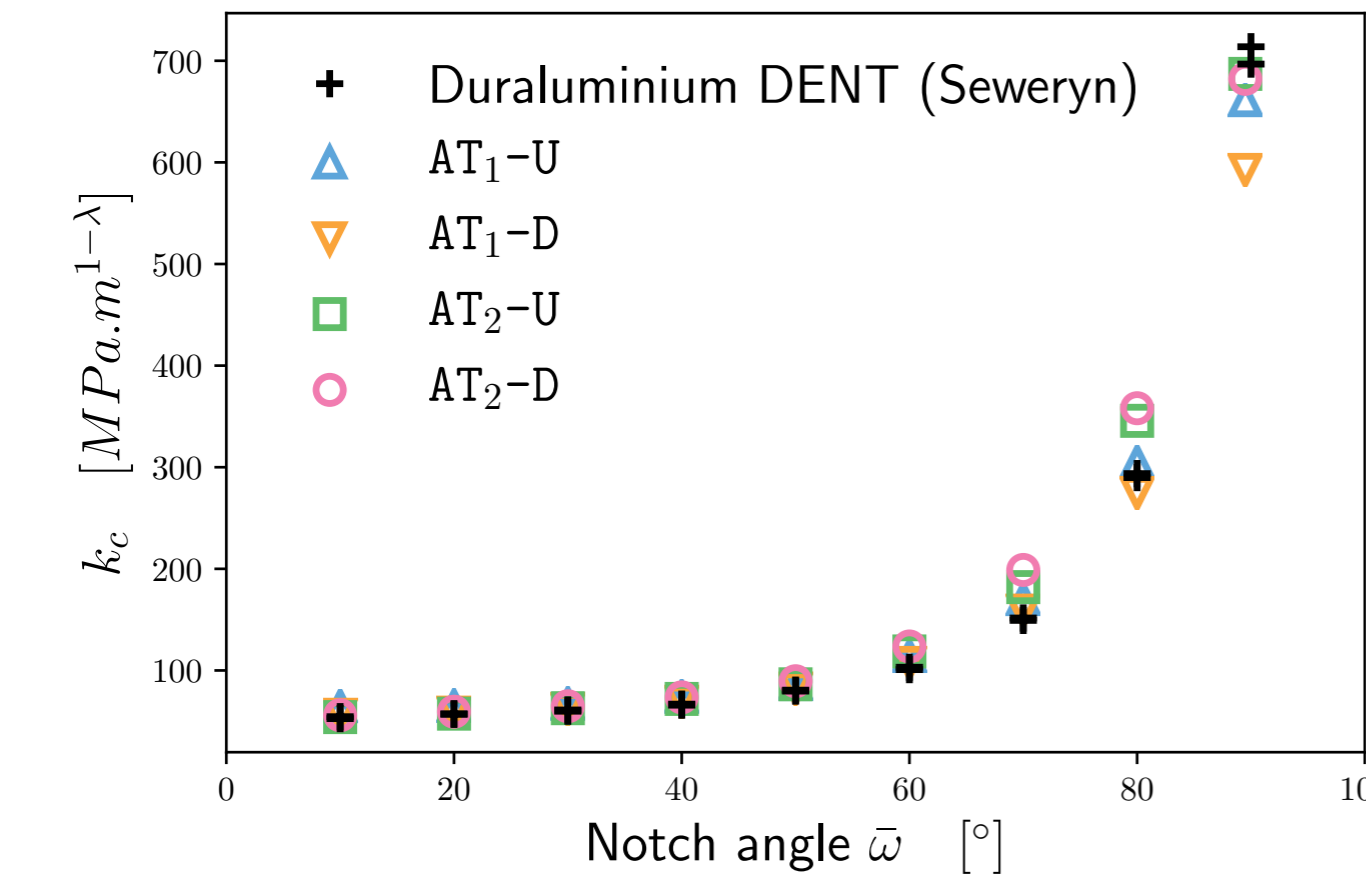
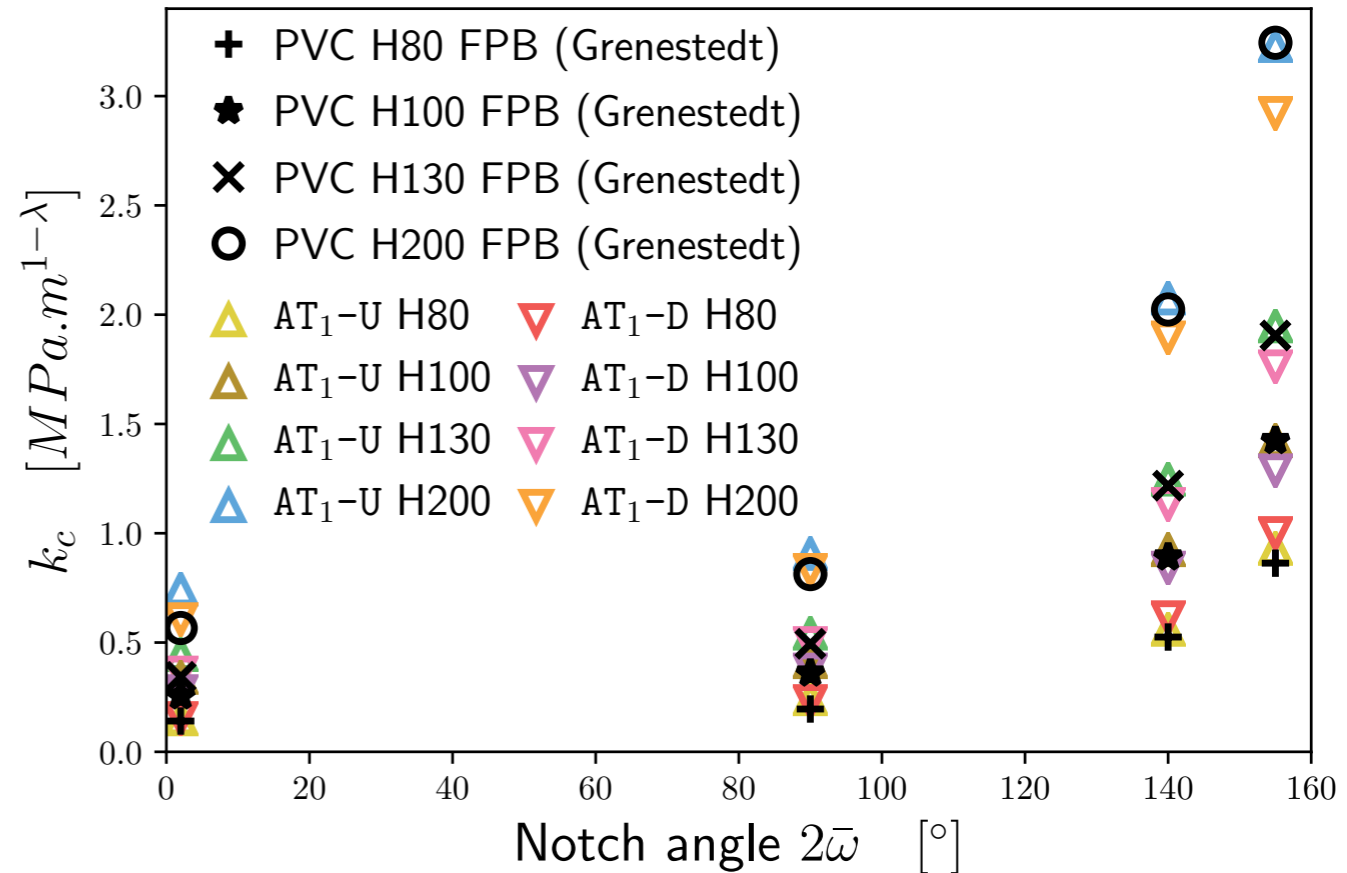
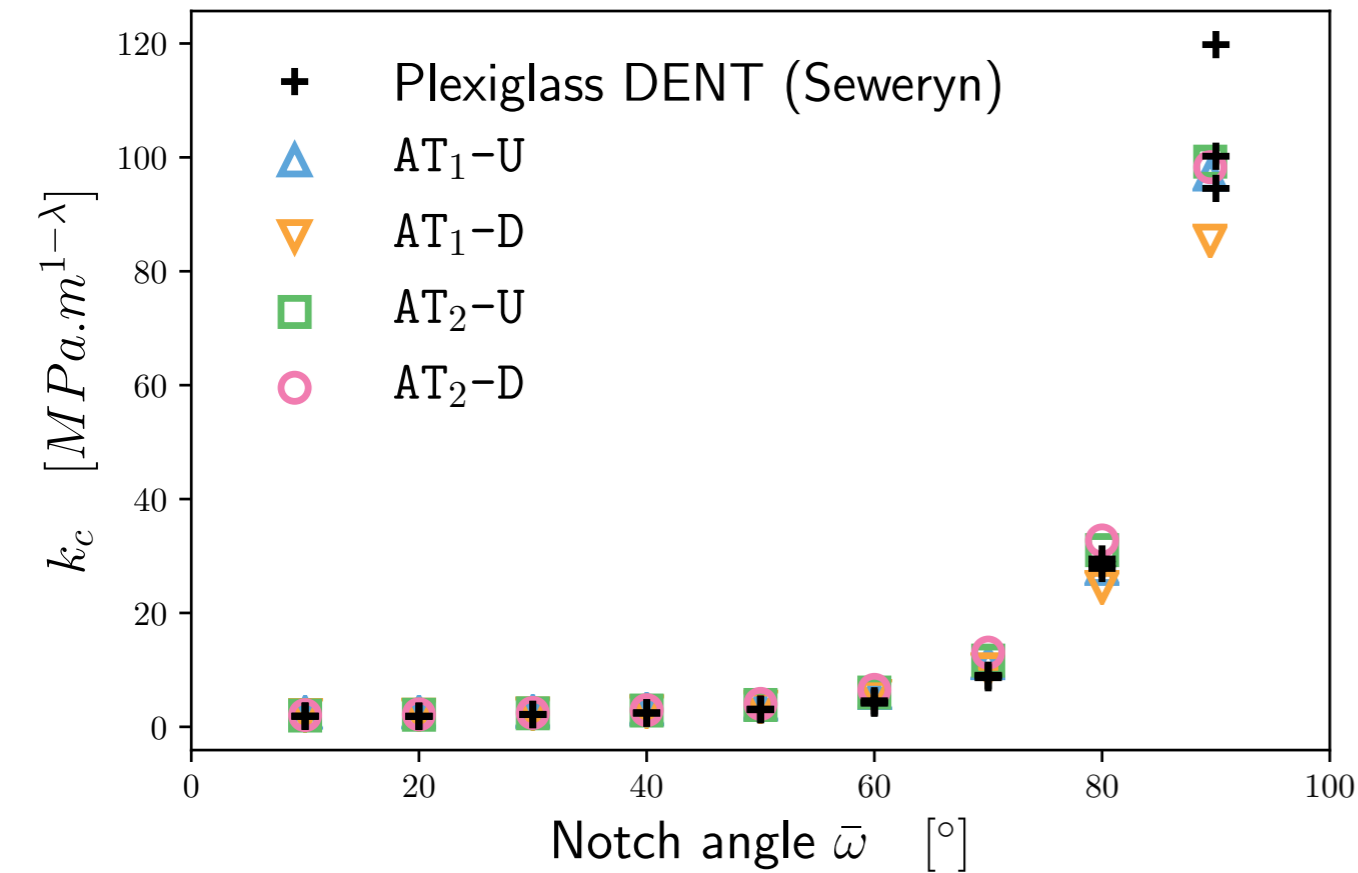
Effect of internal length:



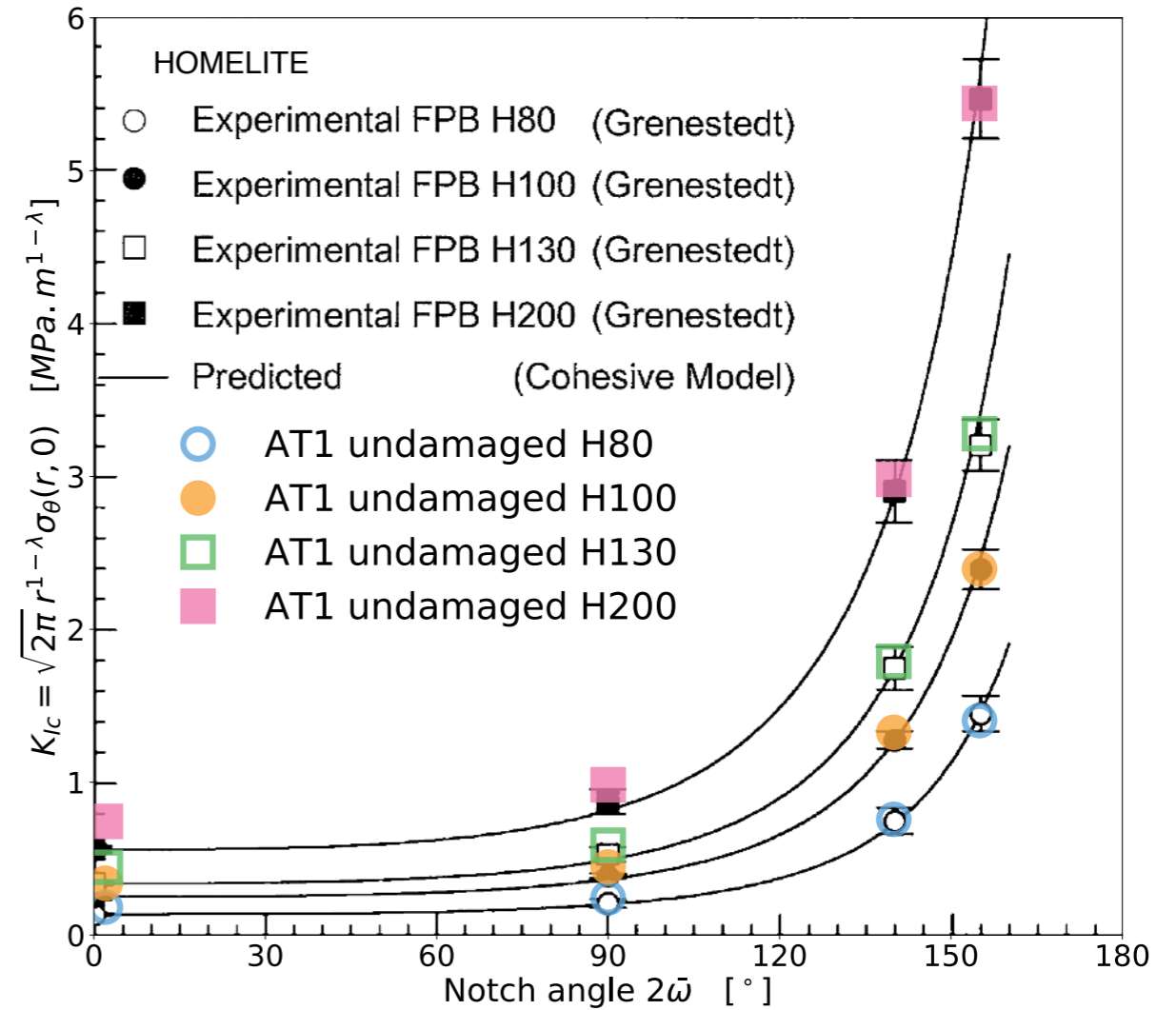
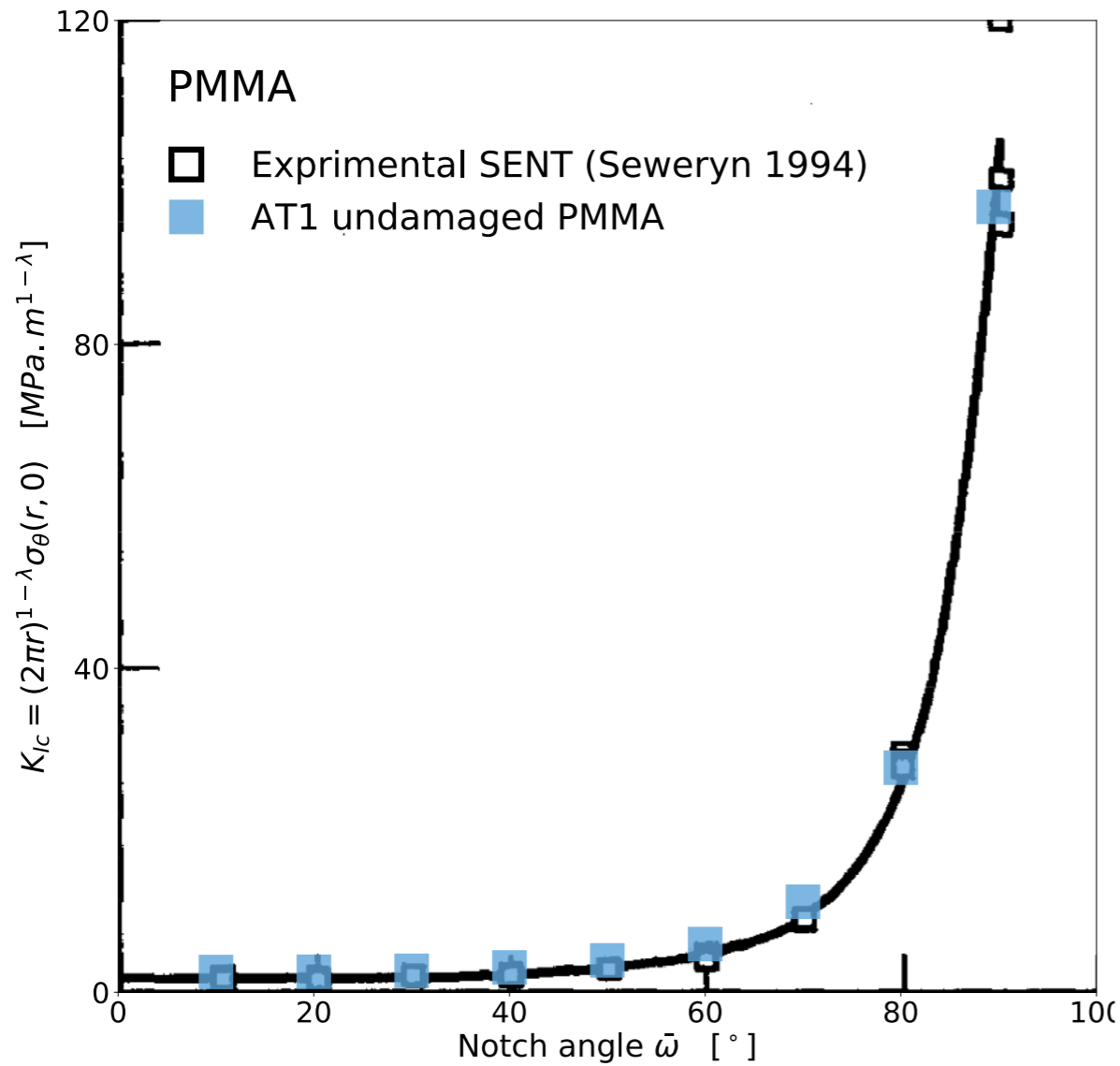
Critical generalized SIF vs notch angle  $k = \sigma_{yy}/(2\pi r)^{\lambda-1}$



# Validation: V-notch, GSIF



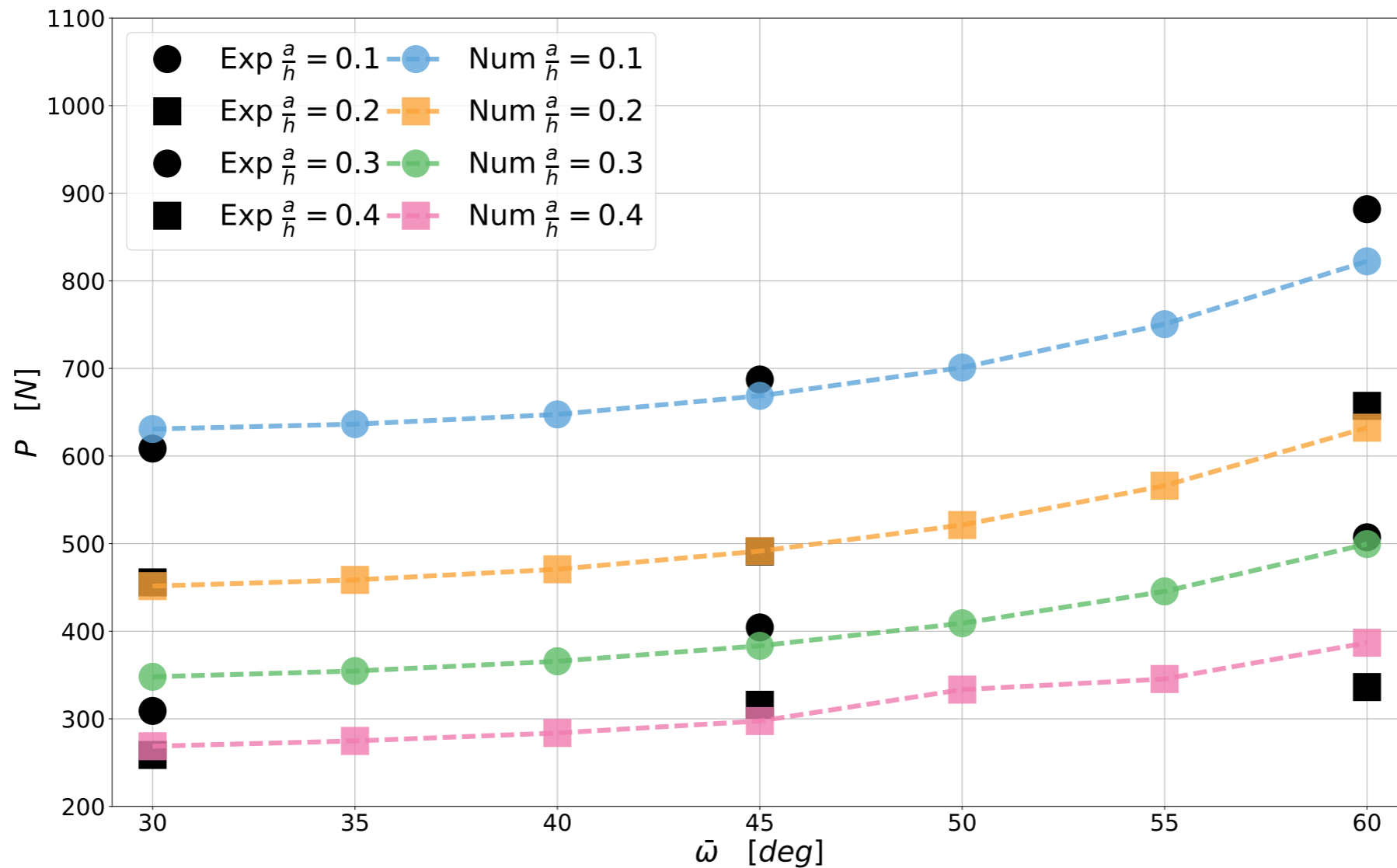
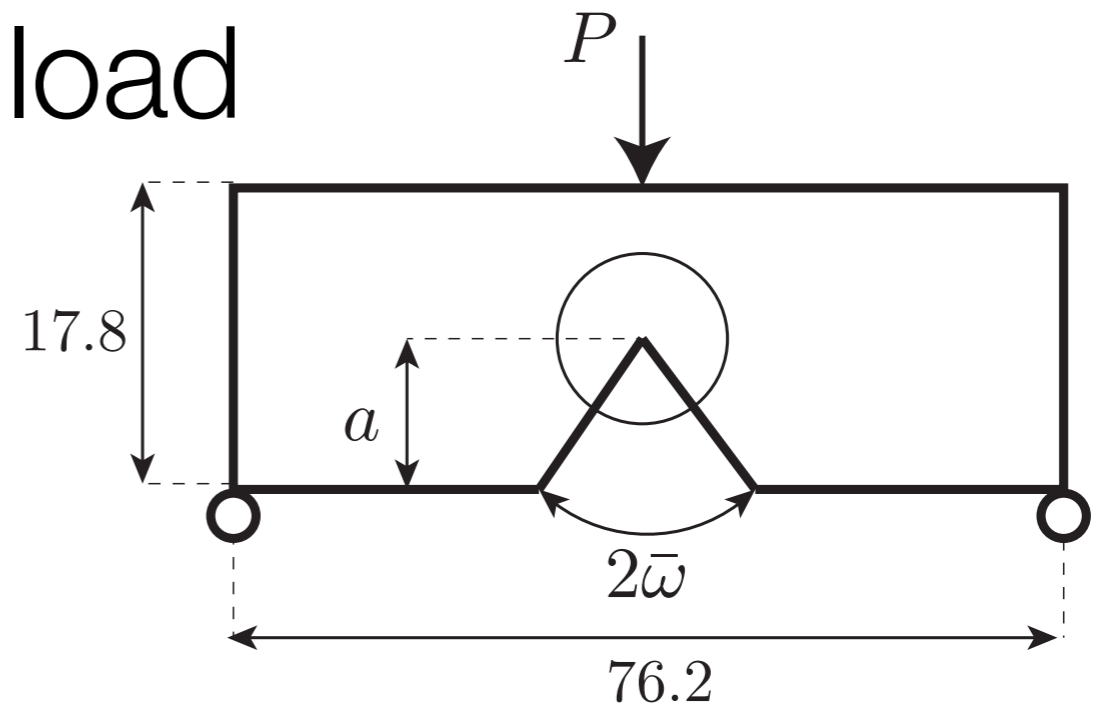
# Validation: V-notch, GSIF



V-notch, cGSIF vs Gómez-Elices *IJF* '06, Seweryn *EFM* '94



# Validation: V-notch, crit. load



V-notch, TPB, PMMA. Failure load vs. Yosibash et al '04

# Current state of phase-field fracture models

Dual view of Variational Phase-Field fracture models.

- Regularization method for the variational approach to fracture:
  - Crack *propagation* insensitive to  $a, w$  and to some extent  $\ell$ .
  - Strong influence of  $a, w, \ell$  on nucleation.
- Gradient-damage models with  $\ell = \ell_{\text{ch}} = \mathcal{O}(K_{Ic}^2 / \sigma_c^2)$ 
  - Effect of  $a, w$  on nucleation can be understood by stability analysis.

*Predictive, validated theory* for propagation *and* nucleation in brittle materials (see also Pham-Ravi-Chandar-Landis, *IJF*, '17).

Variational phase field models of fracture provide a unified way to recover well-accepted propagation and nucleation criteria from a variational model. They involve the same characteristic length as cohesive laws while remaining faithful to Griffith's postulates.

# Toughening in the brittle regime

Commonly accepted toughening mechanisms:

- Crack path deviation, bridging;
- Dissipation by microscopic cracks;
- Plastic dissipation, ductile effects.

What is toughening, “effective toughness” of heterogeneous materials?

- Giacomini '05: elastic and fracture properties homogenize separately, effective toughness through  $\Gamma$ -limit.
- Hossain-Hsueh-Bhattacharya-B *JMPS* '14: mechanism to compute effective toughness:
  - Surfing BC at *macroscopic* scale, free path at *microstructure* scale.
  - Compute the peaks of the ERR (driving force to obtain steady state).

# V&V: "Bittencourt's door"

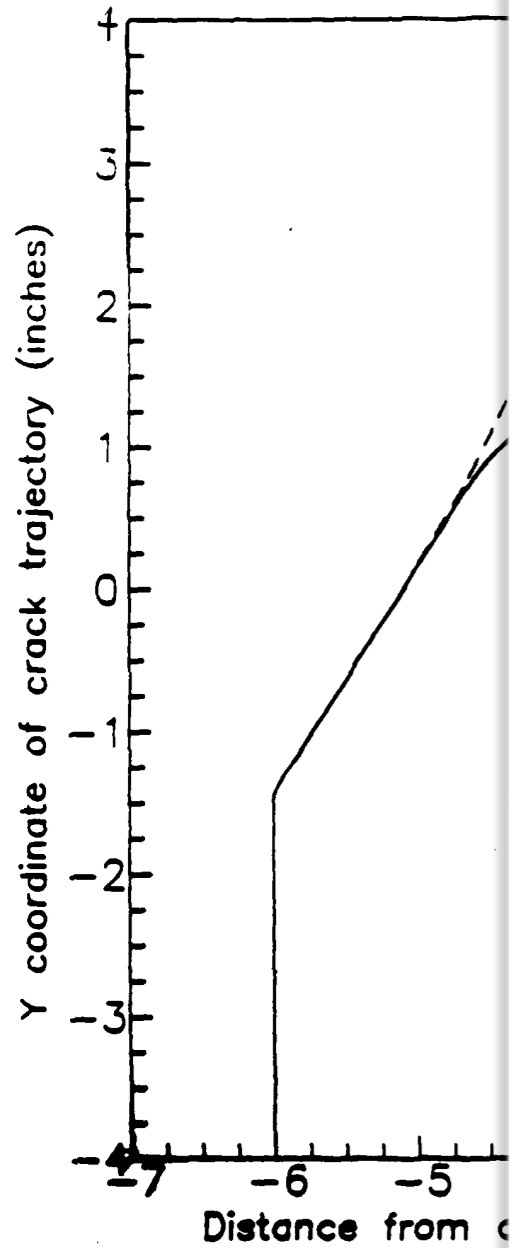


Figure 5.6 Plots of observed crack trajectory with and without

PROBABILISTIC FRACTURE MECHANICS  
A VALIDATION OF PREDICTIVE CAPABILITY

Final Report for AFOSR Project  
Contract # F49620-87-C-0054

Dr. Spencer Wu, Project Monitor

Drs. Anthony R. Ingraffea  
and Mircea Grigoriu  
Principal Investigators

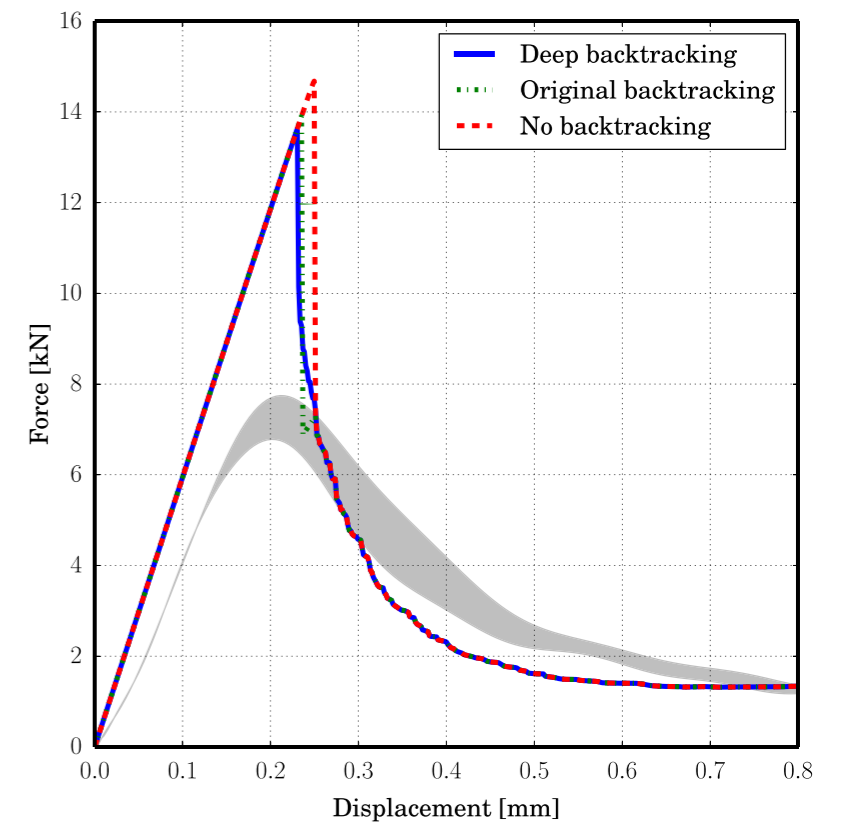
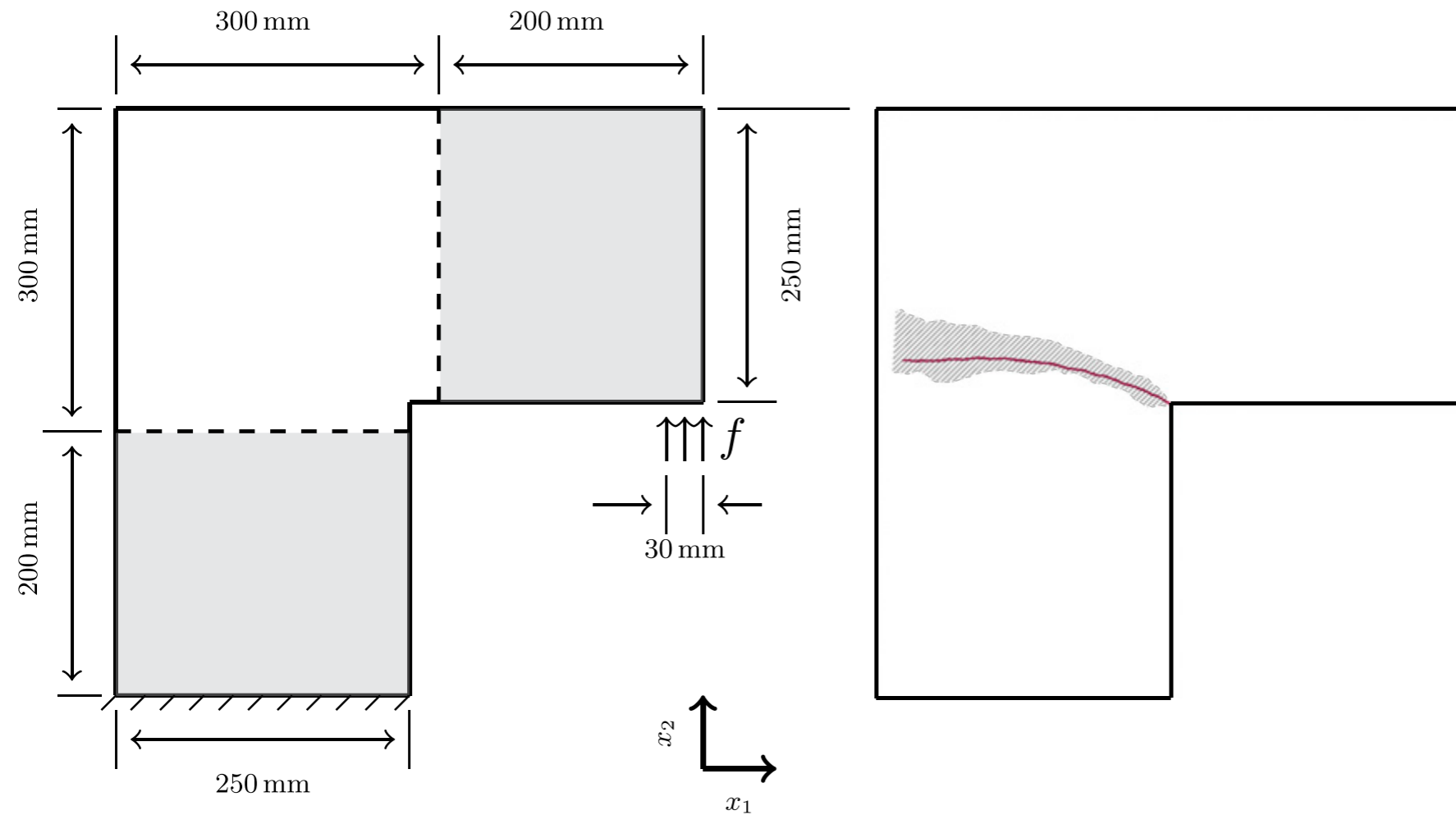
August 1990 Report 90-8

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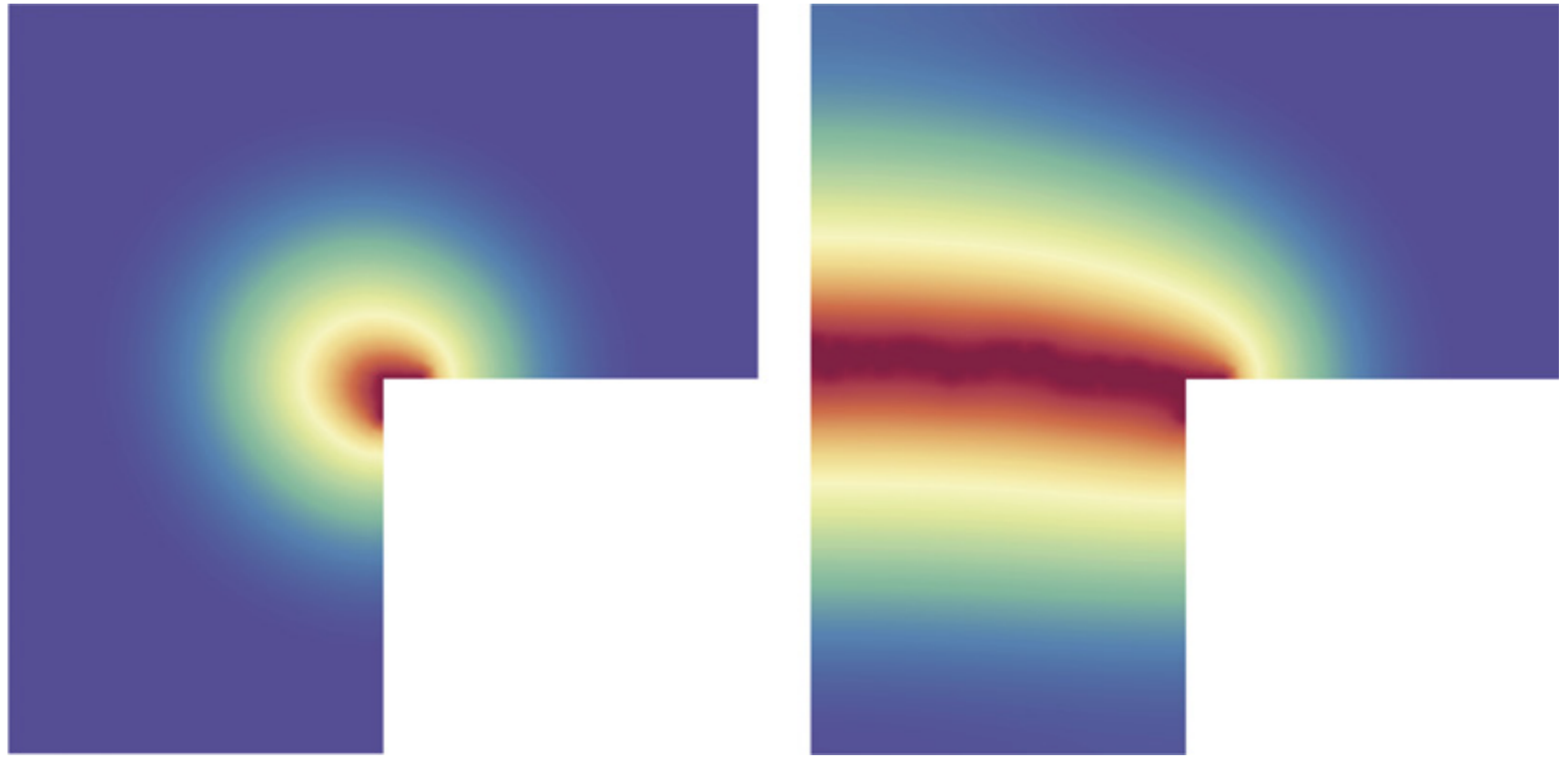
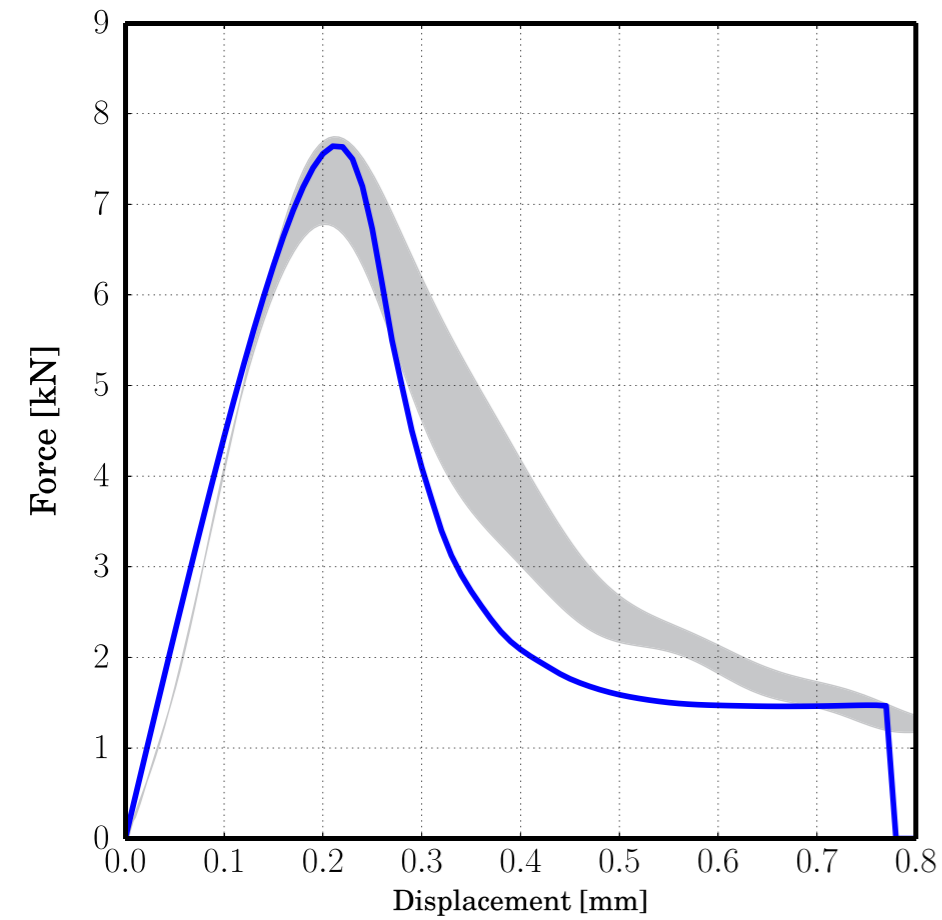
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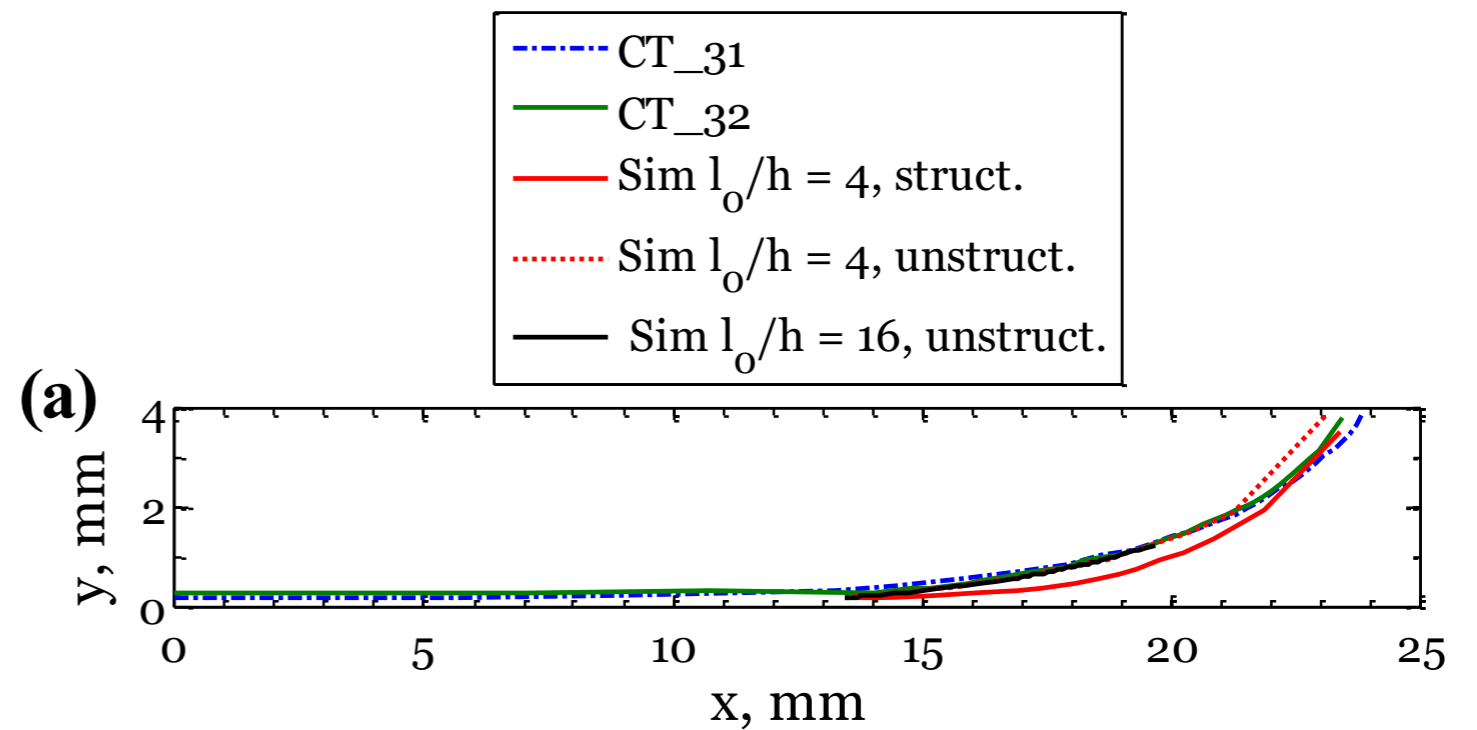
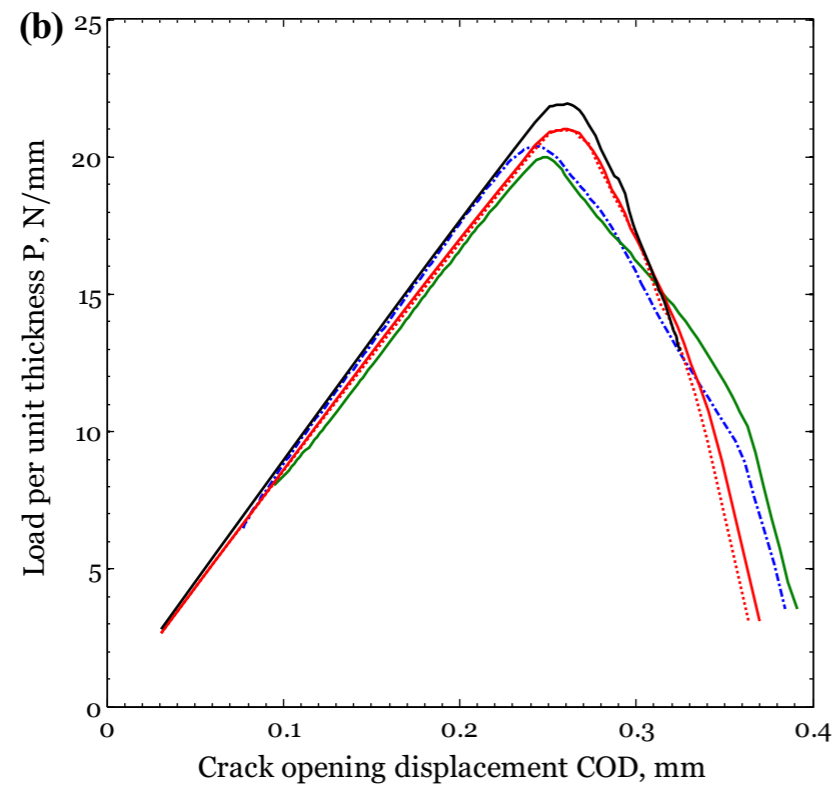
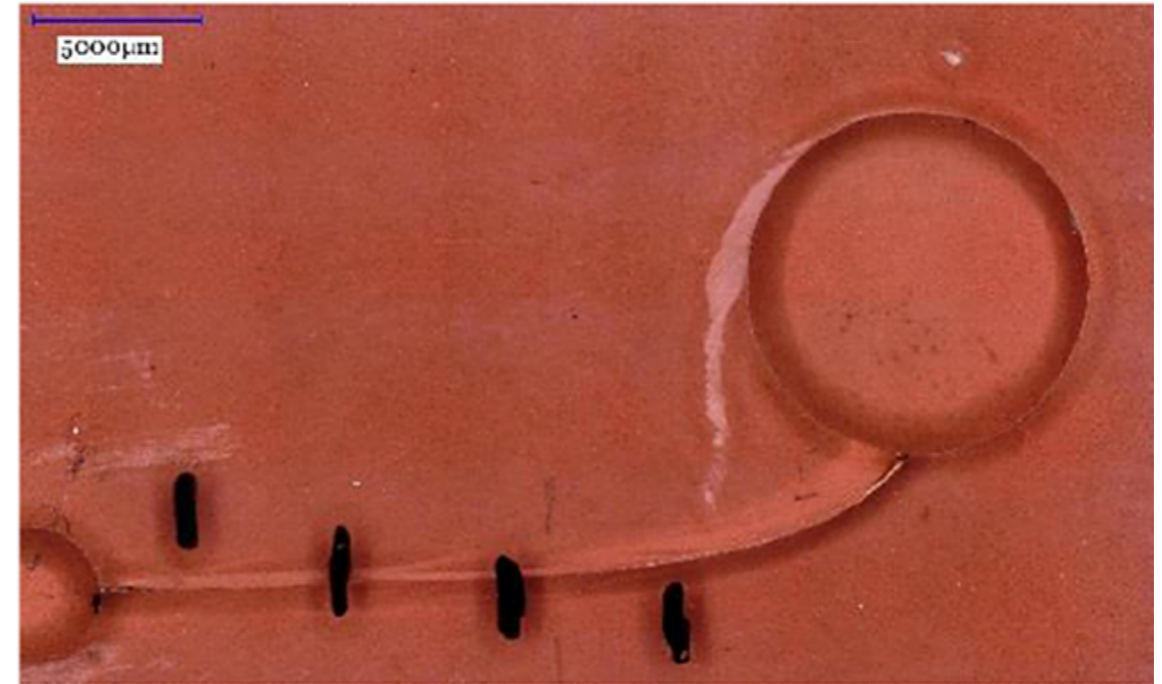
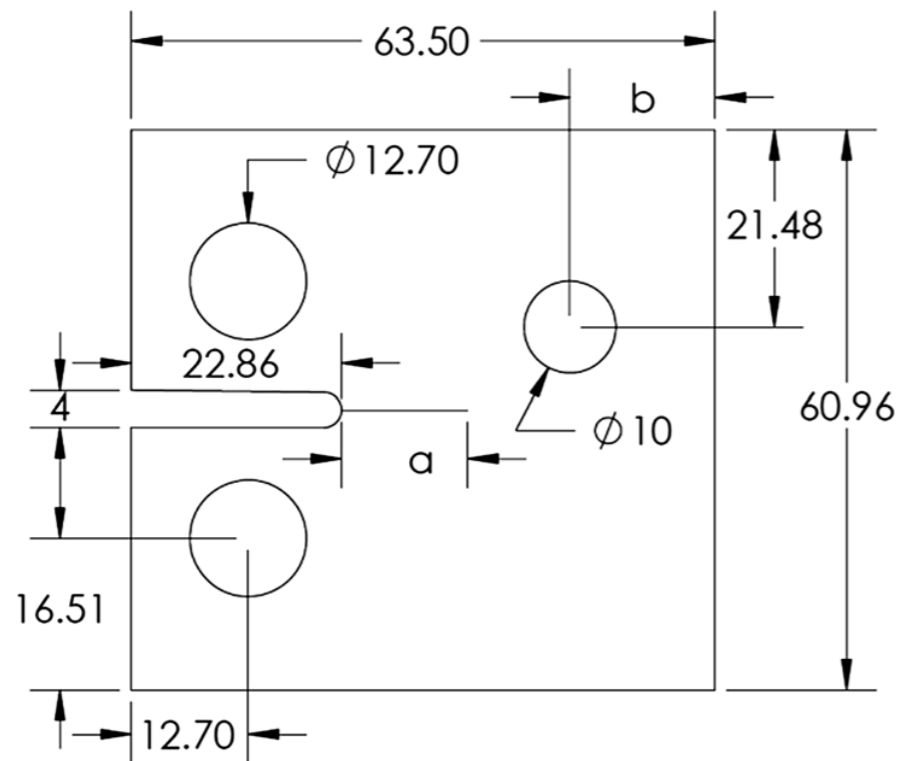
# V&V: L-shaped plate



# V&V: L-shaped plate

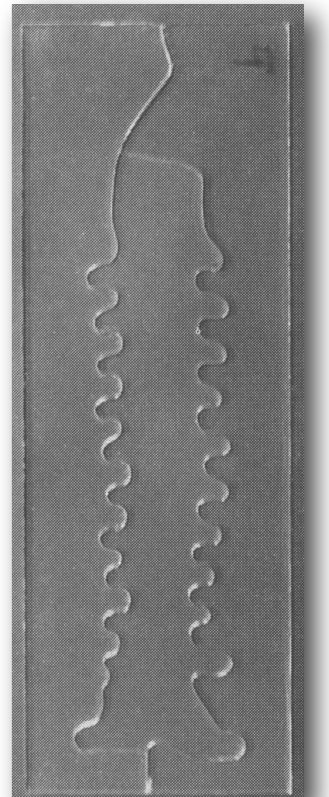
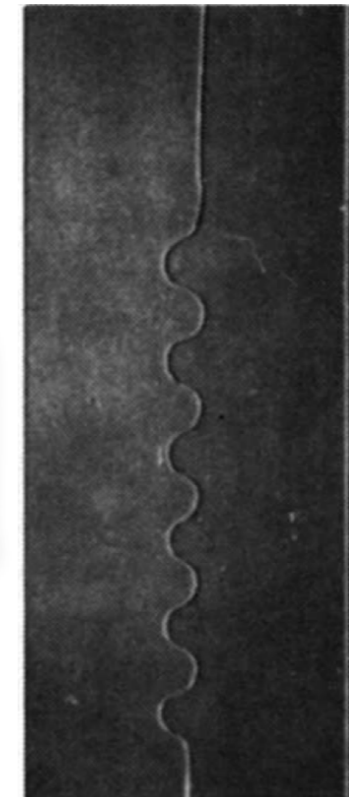
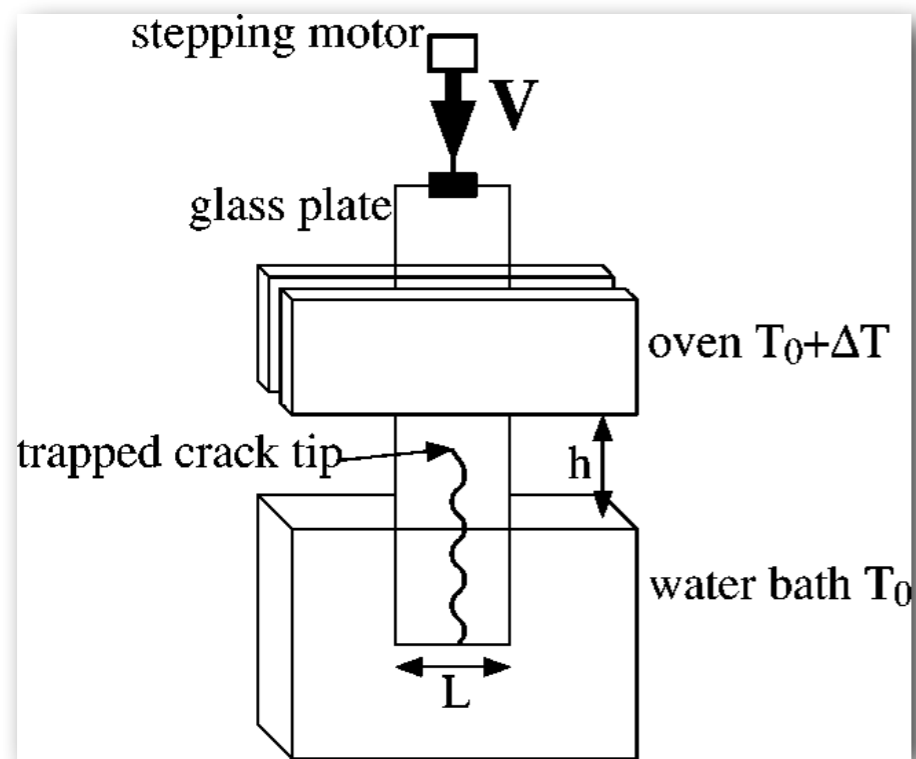
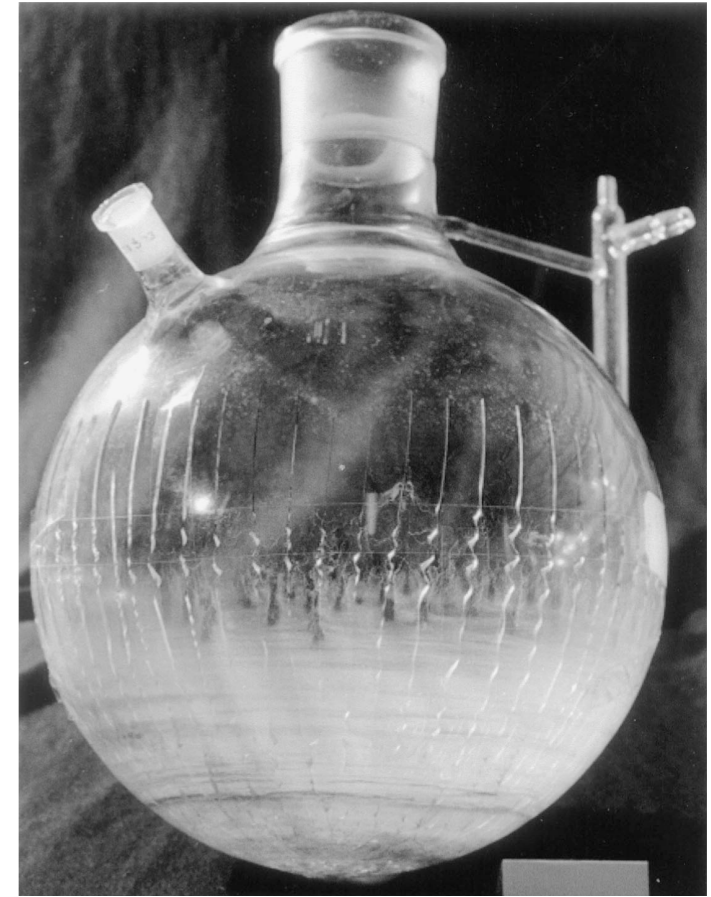


# V&V: Mixed-mode Compact-Tension



# Yuse-Sano experiment

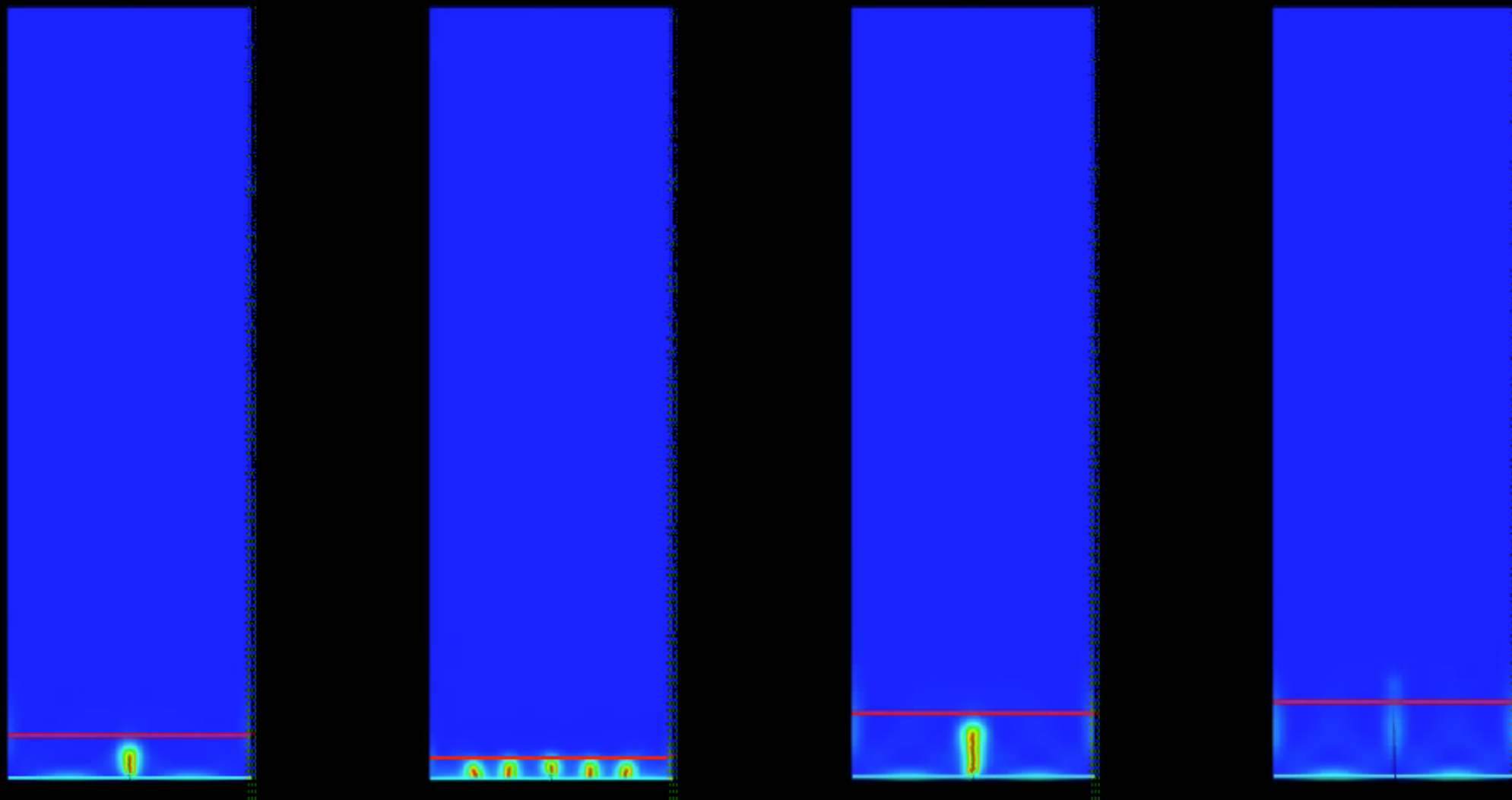
- Glass slabs heated up then quenched in cold water.
- 2 parameters: shock intensity  $\Delta\theta$ , quenching speed  $V$
- 3 regimes: straight, oscillating, erratic.
- *Limited sources of experimental data.*





# Numerical simulations

- Quantitative comparison difficult without direct access to material properties and experimental data.
- Impact of nucleation, heat transfer on simulations.
- Impact of crack geometry on heat transfer?



# Issues and proposed benchmarks

- Fracture community is focused on V&V, not benchmarks
  - Verification: lots of closed form solutions, not all applicable (SIF computations near configurations that cannot be attained!)
  - Validation: good experiments are hard, experimental literature can be sloppy, materials not well quantified.
- Benchmark problems are skewed towards specific class of models (few with mixed mode, complex path, ...)
  - Surfing problem (propagation, ERR, dependance on curvature)
  - Uniaxial tension (elastic domain, critical nucleation stress, closed form solution.
  - Pacman (nucleation with varying singularity), lots of experimental data.
  - Mixed mode, complex path, renucleation?
- Performance comparison: is it enough to upload benchmarks? How about VM / containers and data files?